Multi-Valley Superconductivity in Ion-Gated MoS₂ Layers

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ABSTRACT: Layers of transition metal dichalcogenides (TMDs) combine the enhanced effects of correlations associated with the two-dimensional limit with electrostatic control over their phase transitions by means of an electric field. Several semiconducting TMDs, such as MoS₂, develop superconductivity (SC) at their surface when doped with an electrostatic field, but the mechanism is still debated. It is often assumed that Cooper pairs reside only in the two electron pockets at the K/K' points of the Brillouin Zone. However, experimental and theoretical results suggest that a multivalley Fermi surface (FS) is associated with the SC state, involving six electron pockets at Q/Q'. Here, we perform low-temperature transport measurements in ion-gated MoS₂ flakes. We show that a fully multivalley FS is associated



with the SC onset. The Q/Q' valleys fill for doping $\geq 2 \times 10^{13}$ cm⁻², and the SC transition does not appear until the Fermi level crosses both spin-orbit split sub-bands Q_1 and Q_2 . The SC state is associated with the FS connectivity and promoted by a Lifshitz transition due to the simultaneous population of multiple electron pockets. This FS topology will serve as a guideline in the quest for new superconductors.

KEYWORDS: Transition metal dichalcogenides, ionic gating, superconductivity, electron-phonon coupling, Raman spectroscopy, Lifshitz transitions

ransition metal dichalcogenides (TMDs) are layered L materials with a range of electronic properties. Depending on chemical composition, crystalline structure, number of layers (N), doping, and strain, different TMDs can be semiconducting, metallic, and superconducting.¹ Among semiconducting TMDs, MoS₂, MoSe₂, WS₂, and WSe₂ have sizable bandgaps² in the range $\sim 1-2$ eV. When exfoliated from bulk to single layer (1L), they undergo an indirect-to-direct gap transition,^{2–4} offering a platform for electronic and optoelectronic applications,^{1,2,5} such as transistors,^{6–8} photo-detectors,^{9–12} modulators¹³ and electroluminescent devices.^{14,15}

For all TMDs with 2H crystal structure, the hexagonal Brillouin Zone (BZ) features high-symmetry points Γ , M, K and $K'^{4,16}$ (Figure 1a). The minima of the conduction band fall at K, K', as well as at Q, Q', approximately halfway along the Γ -K(K') directions^{4,16} (Figure 1a). In the absence of an outof-plane electric field, the relative position of Q and Q' depends on N and strain.^{4,16,20} The global minimum of the conduction band⁴ sits at K/K' in 1L-MoS₂ and at Q/Q' in few layer (FL)-MoS₂ with $N \ge 4$. When an electric field is applied perpendicular to the MoS₂ plane, inversion symmetry is broken, and the global minimum of the conduction band is shifted to K/K' in any FL-MoS₂¹⁶ (Figures 1b-d). The valleys at K/K' and at Q/Q' are characterized by a different electronphonon coupling (EPC)²¹ and, when inversion symmetry is broken, by a different spin-orbit coupling (SOC).²² In particular, both EPC and SOC are larger in the Q/Q'valleys.^{21,22}

The field-effect transistor (FET) architecture is ideally suited to control the electronic properties of FL flakes, as it simultaneously provides electrostatic control of the transverse electric field and carrier density. In the electric-double-layer (EDL) technique,²³ the standard solid gate dielectric is replaced by an ionic medium, such as an ionic liquid or electrolyte. In this configuration, the EDL that forms at the ionic liquid/electrode interfaces supports electric fields in excess of²⁴ ~10MV/cm, corresponding to surface carrier densities $n_{2d} \gtrsim 10^{14} \text{ cm}^{-2.24}$ Ionic-liquid gating has been used to tune the Fermi level, $E_{\rm F}$, in TMDs and explore transport at different carrier concentrations.²⁵⁻²⁹ The vibrational properties of TMDs can also be controlled by means of the EDL technique, as suggested by gate-induced softening of Ramanactive modes in 1L-MoS $_{2^{\prime}}^{30}$ while the opposite is observed in gated $1L^{31}$ and two-layer $(2L)^{32}$ graphene. Reference 26 reported a gate-induced superconducting state at the surface of liquid-gated MoS₂ flakes with $N \gtrsim 25$,²⁶ while ref 33 detected this down to N = 1.

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Figure 1. (a) BZ for the 2H crystal structure. High symmetry points Γ , M, K, K', and points Q and Q' are indicated. (b–d) 3L-MoS₂ band structure for three doping values. The bands are adapted from ref 16, and were obtained by density functional theory (DFT) calculations using the Quantum ESPRESSO package¹⁷ in the field-effect transistor configuration.¹⁸ The valleys at K/K' have two spin–orbit split sub-bands, with splitting much smaller than at Q/Q', not seen in this scale. (e) Schematic dependence of σ_{2d} for increasing E_F above the global energy minimum of the conduction band, E_0 , at T = 0, for two ratios of the effective masses in the K/K' and Q/Q' electron pockets ($m_{K/K'}$ and $m_{Q/Q'}$). Curves are calculated using eq 6 of ref 19 and setting a total degeneracy of 4 for the K/K' pockets, 6 for the first Q/Q' pocket (Q₁), and 6 for the second Q/Q' pocket (Q₂); physical constants and energy separations are set to unity.

Most of these results have been interpreted in terms of the population of the conduction band minima at K/K', ${}^{26,34-36}$ which are global minima in both 1L-MoS₂, 3,21 and electrostatically doped FL-MoS₂, 16,26,34,35 Figure 1b. Theoretical investigations, however, suggested that the population of the highenergy minima at Q/Q' may have an important role in determining the properties of gated MoS_2 flakes, by providing contributions both to $EPC^{16,21}$ and $SOC.^{22,37}$ Reference 21 predicted that when the Q/Q' valleys of 1L-MoS₂ are populated (Figure 1c,d), EPC strongly increases (from ~0.1 to ~ 18), leading to a superconducting transition temperature $T_{\rm c} \sim 20 {\rm K}$ for a doping level $x = 0.18 {\rm ~electrons}({\rm e}^-)/{\rm unit~cell}$ (corresponding to $E_{\rm F} = 0.18 \pm 0.02$ eV at K/K' and 0.08 \pm 0.02 eV at Q/Q').¹⁶ However, ref 33 measured $T_{\rm c} \sim 2$ K for $x \sim 0.09-0.17$ e⁻/unit cell in e⁻-doped 1L-MoS₂. This mismatch may be associated with the contribution of $e^--e^$ interactions, whose role in the determination of T_c is still under debate.^{38,39} Overall, the agreement between the model of ref 21 and the trend of T_c with e^- doping in ref 26 suggests that the mechanism of ref 21 for EPC enhancement when the Q/Q' valleys are crossed may also hold for FL-MoS₂.

Inversion symmetry can be broken in MoS₂ either by going to the 1L limit,²¹ or by applying a transverse electric field.^{40,41} This leads to a finite SOC,^{40,41} which lifts the spin degeneracy in the conduction band and gives rise to two spin–orbit-split sub-bands in each valley,^{16,40} as shown in Figure 1b–d for FL-MoS₂. When the system is field-effect doped, the inversion symmetry breaking increases with increasing transverse electric field,^{34,36} due to the fact that induced e^- tend to become more localized within the first layer.^{34,36,38} Hence, the SOC and the spin—orbit splitting between the bands increase as well, as was calculated in ref 16.

When combined to the gate-induced SC state,⁴¹ this can give rise to interesting physics, such as spin-valley locking of the Cooper pairs³⁵ and 2d Ising superconductivity (SC)³⁴ with a non-BCS-like energy gap,⁴² suggested to host topologically nontrivial SC states.^{38,43,44} References 16 and 22 predicted SOC and spin-orbit splitting between sub-bands to be significantly stronger for the Q/Q' valleys than for K/K', thus supporting spin-valley locking at Q/Q' as well.³⁷ A dominant contribution of the Q/Q' valleys in the development of the SC state would be consistent with the high (\gtrsim 50T) inplane upper critical field, H_{c2}^{\parallel} , observed in ion-gated MoS₂^{34,35} and WS₂.⁴⁵ The H_{c2}^{\parallel} enhancement is caused by locking of the spin of the Cooper pairs in the out-of-plane direction in a 2d superconductor in the presence of finite SOC, and is therefore promoted by increasing the SOC. However, H_{c2}^{\parallel} for MoS₂ and WS₂ is higher than in metallic TMD Ising superconductors (such as NbSe₂ and TaS₂), where $H_{c2}^{\parallel} \lesssim 30T$,⁴⁶ despite the SOC in the K/K' valleys being much smaller (~3 meV for MoS_2^{40}). Spin-valley locking in the Q/Q' valleys may thus explain this apparent inconsistency in the physics of ion-gated semiconducting TMDs under magnetic field.

From the experimental point of view, the possible multivalley character of transport in gated TMDs is currently debated. References 36,37, and 47 measured the Landau-level degeneracy at moderate $n_{2d} \sim 10^{12}-10^{13}$ cm⁻², finding it compatible with a carrier population in the Q/Q' valleys. However, ref 36 argued that this would be suppressed for larger $n_{2d} \gtrsim 10^{13}$ cm⁻², typical of ion-gated devices and mandatory for the emergence of SC, due to stronger confinement within the first layer.³⁶ In contrast, angle-resolved photoemission spectroscopy in surface-Rb-doped TMDs⁴⁸ highlighted the presence of a non-negligible spectral weight at the Q/Q' valleys only for $n_{2d} \gtrsim 8 \times 10^{13}$ cm⁻² in the case of MoS₂. Thus, which valleys and sub-bands are involved in the gate-induced SC state still demands a satisfactory answer.

Here we report multivalley transport and SC at the surface of liquid-gated FL-MoS₂. We use a dual-gate geometry to tune doping across a wide range of $n_{2d} \sim 5 \times 10^{12} - 1 \times 10^{14} \text{cm}^{-2}$, induce SC, and detect characteristic "kinks" in the transconductance. These are nonmonotonic features that emerge in the n_{2d} -dependence of the low-temperature (T) conductivity when $E_{\rm F}$ crosses the high-energy sub-bands,¹⁹ irrespectively of their specific effective masses, Figure 1e. We show that the population of the Q/Q' valleys is fundamental for the emergence of SC. The crossing of the first sub-band Q1 (Figure 1c) occurs at small $n_{2d} \lesssim 2 \times 10^{13}$ cm⁻², implying that multivalley transport already occurs in the metallic phase over a wide range of $n_{2d} \sim 2-6 \times 10^{13}$ cm⁻². We also show that the crossing of the second sub-band Q_2 occurs after a finite T_c is observed, while a large population of both spin-orbit-split subbands (Figure 1d) in the Q/Q' valleys is required to reach the maximum T_c . These results highlight how SC can be enhanced in MoS₂ by optimizing the connectivity of its Fermi Surface (FS), i.e., by adding extra FSs in different BZ regions to provide coupling to further phonon branches.⁴⁹ Since the evolution of the band structure of MoS2 with field-effect doping is analogous to that of other semiconducting TMDs,^{16,19,37,39,48} a similar mechanism is likely associated with the emergence of SC in TMDs in general. Thus, optimization of the FS connectivity can be a viable strategy in the search of new superconductors.

We study flakes with N = 4-10, as refs 4, 16, and 19 predicted that flakes with $N \ge 4$ are representative of the bulk electronic structure, and ref 33 experimentally observed that both T_c and the critical magnetic field H_{c2} in 4L flakes are similar to those of 6L and bulk flakes. Our devices are thus comparable with those in literature.^{26,33-35,50} We do not consider 1L flakes as they exhibit a lower T_c , and their mobility is suppressed due to disorder.^{33,50}

FL-MoS₂ flakes are prepared by micromechanical cleavage⁵¹ of 2H-MoS₂ crystals from SPI Supplies. The 2H phase is selected to match that in previous reports of gate-induced SC.^{26,33} Low resistivity (<0.005Ω·cm) Si coated with a thermal oxide SiO₂ is chosen as a substrate. We tested both 90 or 285 nm SiO₂ obtaining identical SC results. Thus, 90 nm SiO₂ is used to minimize the back gate voltage V_{BG} ($-30 \text{ V} < V_{BG} < 30 \text{ V}$), while 285 nm is used to minimize leakage currents through the back gate I_{BG} . Both SiO₂ thicknesses provide optical contrast at visible wavelengths.⁵² A combination of optical contrast, Raman spectroscopy, and atomic force microscopy (AFM) is used to select the flakes and determine N.

Electrodes are then defined by patterning the contacts area by e-beam lithography, followed by Ti:10 nm/Au:50 nm evaporation and lift-off. Ti is used as an adhesion layer,⁵³ while the thicker Au layer provides the electrical contact. Flakes with irregular shapes are further patterned in the shape of Hall bars by using poly(methyl methacrylate) (PMMA) as a mask and removing the unprotected MoS_2 with reactive ion etching (RIE) in a 150 mTorr atmosphere of CF_4 : $O_2 = 5:1$, as shown in Figures 2a,b. A droplet of 1-butyl-1-methylpiperidinium



Figure 2. (a) Hall bar FL-MoS₂ flake with voltage probes (V_i), source (S), drain (D) and liquid-gate (LG) electrodes. A ionic liquid droplet covers the flake and part of the LG electrode. The sample is biased with a source-drain voltage (V_{SD}), and dual gate control is enabled by a voltage applied on the liquid gate (V_{LG}) and on the solid back gate (V_{BG}). (b) Optical image of Hall bar with six voltage probes. The LG electrode is on the upper-right corner. (c) AFM scan of the MoS₂ Hall bar after ionic liquid removal.

bis(trifluoromethylsulfonyl)imide (BMPPD-TFSI) is used to cover the FL-MoS₂ surface and part of the side electrode for liquid gate operation (LG), as sketched in Figure 2a.

AFM analysis is performed with a Bruker Dimension Icon in tapping mode. The scan in Figure 2c is done after the low-T experiments and removal of the ionic liquid, and confirms that the FL-MoS₂ sample does not show topographic damage after the measurement cycle.

We use Raman spectroscopy to characterize the devices both before and after fabrication and BMPPD-TFSI deposition. Raman measurements are performed with a Horiba LabRAM Evolution at 514 nm, with a 1800 grooves/mm grating and a spectral resolution ~0.45 cm⁻¹. The power is kept below 300 μ W to avoid any damage. A representative Raman spectrum of 4L-MoS₂ is shown in Figure 3 (blue curve). The peak at ~455



Figure 3. Representative Raman spectra at 514 nm of a 4L-MoS₂ flake before (blue) and after (red) device fabrication, deposition of the ionic liquid droplet and low-*T* transport measurements.

cm⁻¹ is due to a second-order longitudinal acoustic mode at the M point.⁵⁴ The E_{2g}^1 peak at ~385 cm⁻¹ and the A_{1g} at ~409 cm⁻¹ correspond to in-plane and out-of plane vibrations of Mo and S atoms.^{55,56} Their difference, $Pos(E_{2g}^1)$ -Pos (A_{1g}) , is often used to monitor N.⁵⁷ However, for $N \ge 4$, the variation in $Pos(E_{2g}^1)$ -Pos (A_{1g}) between N and N+1 approaches the instrument resolution⁵⁷ and this method is no longer reliable. Thus, we use the low frequency modes (<100 cm⁻¹) to monitor N.^{58,59} The shear (C) and layer breathing modes (LBM) are due to the relative motions of the atomic planes, either perpendicular or parallel to their normal.⁵⁸ Pos(C) and Pos(LBM) change with N as^{58,59}

$$\operatorname{Pos}(C)_{N} = \frac{1}{\sqrt{2}\pi c} \sqrt{\frac{\alpha_{\parallel}}{\mu_{m}}} \sqrt{1 + \cos\left(\frac{\pi}{N}\right)}$$
(1)

$$\operatorname{Pos}(\operatorname{LBM})_{N} = \frac{1}{\sqrt{2}\pi c} \sqrt{\frac{\alpha_{\perp}}{\mu_{m}}} \sqrt{1 - \cos\left(\frac{\pi}{N}\right)}$$
(2)

where $\alpha_{\parallel} \sim 2.82 \times 10^{19}$ N/m³ and $\alpha_{\perp} \sim 8.90 \times 10^{19}$ N/m³ are spring constants for C and LBM modes, respectively, *c* is the speed of light in vacuum, and $\mu_{\rm m} \sim 3 \times 10^{-6}$ kg/m² is the 1L mass per unit area. ^{58,59} Figure 3 shows a C mode at ~30 cm⁻¹ and an LBM at ~22 cm⁻¹. These correspond to N = 4 using eqs 1 and 2. Figure 3 also plots the Raman measurements after device fabrication, deposition of the ionic liquid, low-*T* measurements, $V_{\rm LG}$ removal and warm-up to room *T* (red curve). We still find Pos(C) ~ 30 cm⁻¹ and Pos(LBM) ~ 22 cm⁻¹, the same as those of the pristine flake, suggesting no damage nor residual doping.

Four-probe resistance and Hall measurements are then performed in the vacuum chambers of either a Cryomech pulse-tube cryocooler, $T_{min} = 2.7$ K, or a Lakeshore cryogenic probe-station, $T_{min} = 8$ K, equipped with a 2T superconducting magnet. A small (~1 μ A) constant current is applied between

S and D (Figure 2a) by using a two-channel Agilent B2912A source-measure unit (SMU). The longitudinal and transverse voltage drops are measured with an Agilent 34420 low-noise nanovoltmeter. Thermoelectrical and other offset voltages are eliminated by measuring each resistance value and inverting the source current in each measurement.⁶⁰ Gate biases are applied between the corresponding G and D with the same two-channel SMU (liquid gate) or a Keithley 2410 SMU (back gate). Samples are allowed to degas in vacuum (<10⁻⁵mbar) at room T for at least ~1 h before measurements, in order to remove residual water traces in the electrolyte.

We first characterize the *T* dependence of the sheet resistance, $R_{\rm s}$, under the effect of the liquid top gate. We apply the liquid gate voltage, $V_{\rm LG}$, at 240 K, where the electrolyte is still liquid, and under high-vacuum (<10⁻⁵mbar) to minimize unwanted electrochemical interactions and extend the stability window of the ionic liquid.²⁴ After $V_{\rm LG}$ is applied, we allow the ion dynamics to settle for ~10 min before cooling to a base T = 2.7 K.

Figure 4a plots the T dependence of R_s measured in a fourprobe configuration, for different V_{LG} and induced carrier density n_{2d} . Our devices behave similarly to ref 26, undergoing first an insulator-to-metal transition near $R_s \sim h/e^2$ at low $n_{2d} <$ 1×10^{13} cm⁻², followed by a metal-to-superconductor transition at high $n_{2d} > 6 \times 10^{13}$ cm⁻². The saturating behavior in the $R_{\rm s}$ versus *T* curves in Figure 4a for $T \lesssim 50$ K, close to the insulator-to-metal transition, is typically observed in systems at low n_{2d} characterized by a fluctuating electrostatic potential, such as that due to charged impurities.⁶¹ This applies to iongated crystalline systems at low $V_{\rm LG}$, since the doping is provided by a low density of ions in close proximity to the active channel. These ions induce a perturbation of the local electrostatic potential, locally inducing charge carriers, but are otherwise far apart. The resulting potential landscape is thus inhomogeneous. This low-doping ($\lesssim 1 \times 10^{13} \text{ cm}^{-2}$) density inhomogeneity is a known issue in ion-gated crystalline systems, but becomes less and less relevant at higher ionic densities.⁶² We employ Hall effect measurements to determine n_{2d} as a function of V_{LG} (see Figure 4b), and, consequently, the liquid gate capacitance $C_{\rm LG}$ for the BMPPD-TFSI/MoS $_2$ interface (~3.4 \pm 0.6 μ F/cm²) is of the same order of magnitude as for DEME-TFSI/MoS₂ in ref 63 (\sim 8.6 ± 4.1 μ F/cm²), where DEME-TFSI is the N,N-diethyl-N-methyl-N-(2-methoxyethyl)ammonium bis(trifluoromethanesulfonyl)imide ionic liquid.⁶³

Figure 4a shows that, while for $T \gtrsim 100$ K R_s is a monotonically decreasing function of n_{2d} , the same does not hold for $T \lesssim 100$ K, where the various curves cross. In particular, the residual R_s in the normal state R_s^0 (measured just above T_c when the flake is superconducting) varies non-monotonically as a function of n_{2d} . This implies the existence of multiple local maxima in the $R_s^0(n_{2d})$ curve. Consistently with the theoretical predictions of ref 19, we find two local maxima. The first and more pronounced occurs when the flake is superconducting, i.e., for $n_{2d} > 6 \times 10^{13}$ cm⁻². This feature was also reported in refs 26 and 34, but not discussed. The second, less pronounced kink, is observed for $1 \times 10^{13} \leq n_{2d} \leq 2 \times 10^{13}$ cm⁻², not previously shown. Both kinks can be seen only for $T \leq 70$ K, and they are smeared for $T \gtrsim 150$ K.

The kink that emerges in the same range of n_{2d} as the superconducting dome extends across a wide range of V_{LG} ($3 \leq V_{LG} \leq 6 \text{ V}$) for $n_{2d} \geq 6 \times 10^{13} \text{ cm}^{-2}$, and can be accessed only by LG biasing, due to the small capacitance of the solid



Figure 4. Transport of dual-gated 4L-MoS₂. (a) R_s as a function of T for different n_{2d} . (b) n_{2d} as a function of V_{LG} as determined via Hall effect measurements, for N = 4, 10. The liquid gate capacitances are obtained by a linear fit of the data. (c) σ_{2d} as a function of V_{LG} at $V_{LG} = 0.9$ V, for different T. Each curve is shifted by 3.333×10^{-5} S. The top scale shows the values of n_{2d} estimated from C_{ox} . Solid (dashed) curves are measured for increasing (decreasing) V_{BG} .

BG. This prevents a continuous characterization of its behavior, as n_{2d} induced by LG cannot be altered for $T \leq 220$ K, as the ions are locked when the electrolyte is frozen. The kink that appears early in the metallic state, on the other hand, extends across a small range of n_{2d} ($1 \times 10^{13} \leq n_{2d} \leq 2 \times 10^{13}$ cm⁻²), and is ideally suited to be explored continuously by exploiting the dual-gate configuration.

We thus bias our samples in the low-density range of the metallic state $(n_{2d} \sim 7 \times 10^{12} \text{ cm}^{-2})$ by applying $V_{\text{LG}} = 0.9 \text{ V}$, and cool the system to 2.7 K. We then apply V_{BG} and fine-tune n_{2d} across the kink. We constantly monitor I_{BG} to avoid dielectric breakdown. Figure 4c plots σ_{2d} of a representative device subject to multiple V_{BG} sweeps, as n_{2d} is tuned across the kink. This reproduces well the behavior observed for low $V_{\text{LG}} (1 \times 10^{13} \leq n_{2d} \leq 2 \times 10^{13} \text{ cm}^{-2})$. The hysteresis between increasing and decreasing V_{BG} is minimal. This kink is suppressed by increasing T, similar to LG gating.

 $V_{\rm BG}$ provides us an independent tool to estimate n_{2d} : If $V_{\rm LG}$ is small enough ($V_{\rm LG} \leq 1$ V) so that conduction in the channel can be switched off by sufficiently large negative $V_{\rm BG}$ ($V_{\rm BG} \leq$ -25 V), we can write $n_{2d} = C_{\rm ox}/e \cdot (V_{\rm BG}-V_{\rm th})$. Here, $C_{\rm ox} =$ $\epsilon_{\rm ox}/d_{\rm ox}$ is the back gate oxide specific capacitance, $e = 1.602 \times$ 10^{-19} C is the elementary charge, and $V_{\rm th}$ is the threshold voltage required to observe a finite conductivity in the device. We neglect the quantum capacitance $C_{\rm q}$ of MoS₂, since $C_{\rm q} \gtrsim$ $100 \ \mu {\rm F/cm}^2 \gg C_{\rm ox}$.¹⁶ By using the dielectric constant of SiO₂ $\epsilon_{\rm ox} = 3.9^{64}$ and an oxide thickness $t_{\rm ox} = 90$ nm (or $t_{\rm ox} = 285$ nm, depending on the experiment), we obtain the n_{2d} scale in the top axis of Figure 4c, in good agreement with the corresponding values in Figure 4a, estimated from the Hall effect measurements in Figure 4b.

The bandstructure of field-effect doped NL-MoS₂ depends on N¹⁶ and strain.¹⁹ A fully relaxed N-layer flake, with $N \ge 3$, has been predicted to behave as follows:^{16,19} For small doping $(x \le 0.05e^-/\text{unit cell}$, Figures 1b and 5a), only the two spin– orbit split sub-bands at K/K' are populated. At intermediate doping $(0.05 \le x \le 0.1 \text{ e}^-/\text{unit cell}$, Figures 1c and 5b), E_F crosses the first spin–orbit split sub-band at Q/Q' (labeled Q₁). For large doping $(x \ge 0.1 \text{ e}^-/\text{unit cell}$, Figures 1d and 5c), E_F crosses the second sub-band (Q_2) , and both valleys become highly populated.¹⁶ Even larger doping $(x \ge 0.35 \text{ e}^-/\text{unit cell})$ eventually shifts the K/K' valleys above E_F .¹⁶

When $E_{\rm F}$ crosses these high-energy sub-bands at Q/Q', sharp kinks are expected to appear in the transconductance of gated FL-MoS₂¹⁹ (see Figure 1e). These are reminiscent of a similar behavior in liquid-gated FL graphene, where their appearance was linked to the opening of interband scattering channels upon the crossing of high-energy sub-bands.^{65–67} Even in the absence of energy-dependent scattering, ref 19 showed that σ_{2d} can be expressed as

$$\sigma_{2d} = e^2 \tau \langle v_{\parallel}^2 \rangle N(E_{\rm F}) \propto e^2 \langle v_{\parallel}^2 \rangle \tag{3}$$

where $\tau \propto N(E_{\rm F})^{-1}$ is the average scattering time, and $N(E_{\rm F})$ is the density of states (DOS) at $E_{\rm F}$. This implies that σ_{2d} is proportional to the average of the squared in-plane velocity $\langle \nu_{\parallel}^2 \rangle$ over the FS.¹⁹ Since $\langle \nu_{\parallel}^2 \rangle$ linearly increases with n_{2d} and

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Figure 5. (a-c) Fermi Surface of 3L-MoS₂ for the three doping values in Figure 1b–d. High symmetry points Γ , M, K, K', and points Q and Q' are shown. Blue arrows indicate representative phonon wave vectors that connect the various FSs. (d) SC dome of liquid-gated MoS₂ as a function of n_{2d} . T_c is determined at 90% of the total transition. (e) R_s^0 as a function of n_{2d} , for increasing V_{LG} (blue filled circles) and V_{BG} (solid red line). In panels d and e, filled circles are our data, and black and magenta open circles are taken from refs 26 and 34. The background is color-coded to indicate the doping ranges highlighted in panels a–c.

drops sharply as soon as a new band starts to get doped,¹⁹ the kinks in σ_{2d} (or, equivalently, R_s) at $T \leq 15$ K can be used to determine the onset of doping of the sub-bands in the Q/Q'valleys. At T = 0, the kink is a sharp drop in σ_{2d} , emerging for the doping value at which $E_{\rm F}$ crosses the bottom of the next sub-band. This correspondence is lost due to thermal broadening for T > 0, leading to a smoother variation in σ_{2d} . If T is sufficiently large, the broadening smears out any signature of the kinks (Figure 4). Reference 19 calculated that, at finite T, the conductivity kinks define a doping range where the sub-band crossing occurs (between R_s minimum and maximum, i.e., the lower and upper bounds of each kink set the resolution of this approach). Each sub-band crossing starts after the R_s minimum at lower doping, then develops in correspondence of the inflection point, and is complete once the R_s maximum is reached.

We show evidence for this behavior in Figure 5, where we plot T_c (panel d) and R_s^0 (panel e) as a function of n_{2d} . The electric field is applied both in liquid-top-gate (filled dots and dashed line) and dual-gate (solid red line) configurations. For comparable values of n_{2d} , the liquid-gated samples reach larger R_s^0 than the back-gated ones. This difference is due to increased disorder introduced when n_{2d} is modulated via ionic gating.⁶⁶⁻⁷⁰ Two kinks appear in the n_{2d} dependence of R_s : a low-doping one for $1.5 \times 10^{13} \leq n_{2d} \leq 2 \times 10^{13}$ cm⁻², and a high-doping one for $7 \times 10^{13} \leq n_{2d} \leq 9 \times 10^{13}$ cm⁻². The plot of the SC dome of gated MoS₂ on the same n_{2d} scale shows that the low-doping kink appears well before the SC onset, while the second appears immediately after, before the maximum T_c is reached.

These results can be interpreted as follows. When $n_{2d} \leq 1 \times 10^{13} \text{ cm}^{-2}$, only the spin-orbit split sub-bands at K/K' are populated, and the FS is composed only by two pockets (Figure 5a). For n_{2d} between ~1.5 and 2 × 10¹³ cm⁻², $E_{\rm F}$

crosses the bottom of the Q₁ sub-band and two extra pockets appear in the FS at Q/Q',^{16,21} Figure 5b. The emergence of these pockets induces a Lifshitz transition, i.e., an abrupt change in the topology of the FS.⁷¹ Once Q₁ is populated and $E_{\rm F}$ is large enough ($n_{2d} \sim 6 \times 10^{13} \, {\rm cm}^{-2}$), the system becomes superconducting.^{26,34} For slightly larger $E_{\rm F}$ (7 × 10¹³ $\leq n_{2d} \leq 9$ × 10¹³ cm⁻²), $E_{\rm F}$ crosses the bottom of Q₂, resulting in a second Lifshitz transition, and two other pockets emerge in the FS at Q/Q¹¹⁶ (Figure 5c).

We note that the experimentally observed kinks are at different n_{2d} with respect to the theoretical ones for 3L-MoS₂.¹⁹ Reference 19 predicted that for a 1.28% in-plane tensile strain, Q₁ and Q₂ should be crossed for $n_{2d} \sim 5 \times 10^{13}$ cm⁻² and $\sim 1 \times 10^{14}$ cm⁻². Since the positions of the sub-band crossings are strongly dependent on strain,¹⁹ we estimate the strain in our devices by monitoring the frequency of the E_{2g}^1 mode via Raman spectroscopy. Strain can arise due to a mismatch in the thermal expansion coefficients (TECs) of MoS₂,⁷² SiO₂ substrate,⁷³ and Au electrodes.⁷⁴ Upon cooling, MoS₂, SiO₂, and Au would normally undergo a contraction. However, the flake is also subject to a tensile strain due to TEC mismatch.⁷⁵ The strain, ϵ_{MoS_2} , due to the MoS₂–SiO₂ TEC mismatch is

$$\epsilon_{\text{MoS}_2} = \int_T^{292\text{K}} \left[\alpha_{\text{MoS}_2}(T) - \alpha_{\text{SiO}_2}(T) \right] dT \tag{4}$$

whereas the strain, $\epsilon_{\rm Au}$ due to the Au contacts is

$$\epsilon_{Au} = \int_{T}^{292K} \left[\alpha_{Au}(T) - \alpha_{SiO_2}(T) \right] dT$$
(5)

 $\epsilon_{\rm MoS_2}$ and $\epsilon_{\rm Au}$ are ~0.1% and ~0.3% at ~4 K, respectively. 75

Any FL-MoS₂ on SiO₂ will be subject to ϵ_{MoS_2} at low T. When the flake is contacted, an additional contribution is present due to ϵ_{Au} . This can be more reliably estimated performing T-dependent Raman scattering and comparing the spectra for contacted and uncontacted flakes.^{75,76} Figure 6a,b shows how a T decrease results in the E_{2g}^1 mode shifting to higher frequencies for both as-prepared and contacted 4L-MoS₂, due to anharmonicity.⁷⁷ However, in the as-prepared 4L-MoS₂, the upshift is $\sim 1 \text{ cm}^{-1}$ larger with respect to the contacted one. This difference points to a further tensile strain. References 78 and 79 suggested that uniaxial tensile strain on IL-MoS₂ induces a E_{2g}^1 softening and a splitting in two components: E_{2g}^{1+} and $E_{2g}^{1-,78,79}$ The shift rates for E_{2g}^{1+} and E_{2g}^{1-} range from -0.9 to -1.0 cm⁻¹/% and from -4.0 to -4.5 cm⁻¹/%, respectively.^{78,79} We do not observe splitting, pointing toward a biaxial strain. As for ref 76, we calculate a shift rate of E_{2g}^1 for biaxial strain ranging from -7.2 to -8.2 cm^{-1} /%. The amount of tensile strain on the 4L-MoS₂ device can thus be estimated. The E_{2g}^1 upshift difference between contacted and as-prepared 4L-MoS₂, $\Delta Pos(E_{2g}^1)$, at 4 K is ~ -1.0 cm⁻¹, corresponding to an additional ~0.13% biaxial tensile strain. Thus, assuming a 0.1% strain for the as-prepared 4L-MoS_2 due to TEC mismatch with $\text{SiO}_2\text{,}$ we estimate the total strain in the contacted 4L-MoS₂ to be ~0.23% at ~4 K.

Figure 7a shows that, for 0.23% tensile strain, the experimentally observed positions of the kinks agree well with a linear extrapolation of the data of ref 19 to 4L-MoS₂ (representative of our experiments) and for in-plane strain between 0% (bulk) and 1.28% (fully relaxed). These findings indicate that, while the mechanism proposed in ref 21 qualitatively describes the general behavior of gated FL-MoS₂, quantitative differences arise due to the spin–orbit split



Figure 6. (a) Raman spectra of the 4L-MoS₂ device in Figure 2c from 4 to 292 K. (b) Shift in the position of the E_{2g}^1 mode as a function of *T* for as-prepared bulk flake (black circles), a 4L-MoS₂ flake (blue circles), and a 4L-MoS₂ device with Au contacts (red circles).

of the Q_1 and Q_2 sub-bands. The main reason for the EPC (and, hence, T_c) increase is the same, i.e., the increase in the number of phonon branches involved in the coupling when the high-energy valleys are populated.²¹ However, the finite spinorbit-split between the sub-bands significantly alters the FS connectivity upon increasing doping.¹⁶ If we consider the relevant phonon wave vectors ($q = \Gamma, K, M, \Gamma K/2$) for 1L- and FL-MoS₂, 80,81 and only the K/K' valleys populated, then only phonons near Γ and K can contribute to EPC.²¹ The former strongly couple e^- within the same valley,²¹ but cannot contribute significantly due to the limited size of the Fermi sheets.²¹ The latter couple e^- across different valleys,²¹ and provide a larger contribution,²¹ insufficient to induce SC. MoS₂ flakes are metallic but not superconducting before the crossing of Q₁. When this crossing happens, the total EPC increases due to the contribution of longitudinal phonon modes near K^{21} (coupling states near two different Q or Q'), near $\Gamma K/2^{21}$ (coupling states near Q to states near Q'), and near M^{21} (coupling states near Q or Q' to states near K or K'). However, this first EPC increase associated with Q1 is not sufficient to induce SC, as the SC transition is not observed until immediately before the crossing of the spin-orbit-split sub-band Q₂ and the second doping-induced Lifshitz transition. Additionally, the SC dome shows a maximum in the increase of T_c with doping (dT_c/dn_{2d}) across the Q_2 crossing, i.e., when a new FS emerges. Consistently, the subsequent reduction of T_c for $n_{2d} \ge 13 \times 10^{13} \text{ cm}^{-2}$ can be



(a) 25

 $n_{2d} \, (10^{13} \, {\rm cm}^{-2})$

(b)

20

15

10

5

0

0.7

0.6

⊲ 0.5

Letter



band in FL-MoS₂ as a function of tensile strain. Theoretical values for 1L (black dots and line) and 3L (red triangle and line) from ref 19; values for 4L (green diamonds and line) are by linear extrapolation. Blue diamond is the present experiment. (b) EPC enhancement due to the crossing of the Q₂ sub-band, $\Delta\lambda$, as a function of n_{2d} assuming $\omega_{ln} = 230 \pm 30$ K and $\mu^* = 0.13$.²¹ Filled blue circles are our experiments. Black and magenta open circles are taken from refs 26 and 34. The blue dashed line is a guide to the eye.

associated with the FS shrinkage and disappearance at K/K',^{16,21} and might also be promoted by the formation of an incipient charge density wave^{82,83} (characterized by periodic modulations of the carrier density coupled to a distortion of the lattice structure⁸⁴).

Since the evolution of the bandstructure with doping is similar in several semiconducting $\text{TMDs}_{,}^{16,19,37,39,48}$ this mechanism is likely not restricted to gated MoS₂. The T_c increase in correspondence to a Lifshitz transition is reminiscent of a similar behavior observed in CaFe₂As₂ under pressure,⁸⁵ suggesting this may be a general feature across different classes of materials.

We note that the maximum $T_c \sim 11$ K is reached at $n_{2d} \simeq 12 \times 10^{13}$ cm⁻², as reported in ref 26. This is a doping level larger than any that can be associated with the kink. Thus, the Q₂ sub-band must be highly populated when the maximum T_c is observed. We address this quantitatively with the Allen-Dynes formula,⁸⁶ which describes the dependence of T_c by a numerical approximation of the Eliashberg theory, accurate for materials with a total $\lambda \leq 1.5$:⁸⁶

$$T_{\rm c}(n_{2d}) = \frac{\omega_{\rm ln}}{1.2} \exp\left\{\frac{-1.04[1+\lambda(n_{2d})]}{\lambda(n_{2d}) - \mu^*[1+0.62\lambda(n_{2d})]}\right\}$$
(6)

where $\lambda(n_{2d})$ is the total EPC as a function of doping, ω_{ln} is the representative phonon frequency and μ^* is the Coulomb pseudopotential. It is important to evaluate the increase in EPC between the nonsuperconducting region ($n_{2d} \lesssim 6 \times 10^{13}$ cm⁻²) and the superconducting one, i.e., the enhancement in λ due to the crossing of the sub-band at Q₂. $\Delta \lambda = \lambda (T_c) - \lambda (T_c)$ = 0) indicates the EPC increase due to the appearance of e^{-1} pockets at Q₂. By setting $\omega_{ln} = 230 \pm 30$ K and $\mu^* = 0.13$ (as for ref 21), and using eq 6, we find that the limit of $\lambda(T_c)$ for $T_{\rm c} \sim 10^{-6}$ K is ~0.25. The corresponding $\Delta \lambda$ vs n_{2d} dependence is shown in Figure 7b. The crossing at Q₂ results in a maximum $\Delta \lambda = 0.63 \pm 0.1$, with a maximum EPC enhancement of $350 \pm 40\%$ with respect to the nonsuperconducting region. This indicates that the largest contribution to the total EPC, hence to the maximum $T_{\rm c} \sim$ 11 K, is associated with the population of the Q₂ sub-band. This is consistent with the reports of a reduced $T_c \sim 2$ K in 1L- $MoS_{2}^{33,50}$ shown to be superconducting for smaller $n_{2d} \sim 5.5 \times 10^{13} \text{ cm}^{-2}$,⁵⁰ and likely to populate Q₁ only. $n_{2d} \sim 5 \times 10^{13}$ cm^{-2} is also the doping expected for the crossing of Q_1 in 1L- MoS_2 in the presence of a low-T strain similar to that in our 4L-MoS₂ devices (see Figure 7a).

In summary, we exploited the large carrier density modulation provided by ionic gating to explore sub-band population and multivalley transport in MoS₂ layers. We detected two kinks in the conductivity, associated with the doping-induced crossing of the two sub-bands at Q/Q'. By comparing the emergence of these kinks with the doping dependence of T_c , we showed how superconductivity emerges in gated MoS_2 when the Q/Q' valleys are populated, while previous works only considered the filling of K/K'. We highlighted the critical role of the population of the second spin-orbit-split sub-band, Q2, (and the consequent increase of the FS available for EPC) in the appearance of superconductivity and in the large enhancement of T_c and of EPC in the first half of the superconducting dome. Our findings explain the doping dependence of the SC state at the surface of gated FL-MoS₂, and provide key insights for other semiconducting transition metal dichalcogenides.

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Notes

The authors declare no competing financial interest.

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REFERENCES

(1) Ferrari, A. C.; Bonaccorso, F.; Fal'ko, V.; Novoselov, K. S.; Roche, S.; Bøggild, P.; Borini, S.; Koppens, F. H.; Palermo, V.; Pugno, N.; Garrido, J. A.; Sordan, R.; Bianco, A.; Ballerini, L.; Prato, M.; Lidorikis, E.; Kivioja, J.; Marinelli, C.; Ryhänen; Morpurgo, A. F.; Coleman, J. N.; Nicolosi, V.; Colombo, L.; Fert, A.; Garcia-Hernandez, M.; Bachtold, A.; Schneider, G. F.; Guinea, F.; Dekker, C.; Barbone, M.; Sun, Z.; Galiotis, C.; Grigorenko, A. N.; Konstantatos, G.; Kis, A.; Katsnelson, M.; Vandersypen, L.; Loiseau, A.; Morandi, V.; Neumaier, D.; Treossi, E.; Pellegrini, V.; Polini, M.; Tredicucci, M.; Williams, G. M.; Hong, B. H.; Ahn, J.-H.; Kim, J. M.; Zirath, H.; van Wees, B. J.; van der Zant, H.; Occhipinti, L.; Di Matteo, A.; Kinloch, I. A.; Seyller, T.; Quesnel, E.; Feng, X.; Teo, K.; Rupesinghe, N.; Hakonen, P.; Neil, S. R. T.; Tannock, Q.; Löfwander, T.; Kinaret, J. Science and technology roadmap for graphene, related two-dimensional crystals, and hybrid systems. *Nanoscale* 2015, *7*, 4598–4810.

(2) Wang, Q. H.; Kalantar-Zadeh, K.; Kis, A.; Coleman, J. N.; Strano, M. S. Electronics and optoelectronics of two-dimensional transition metal dichalcogenides. *Nat. Nanotechnol.* **2012**, *7*, 699.

(3) Mak, K. F.; Lee, C.; Hone, J.; Shan, J.; Heinz, T. F. Atomically thin MoS_2 : a new direct-gap semiconductor. *Phys. Rev. Lett.* **2010**, 105, 136805.

(4) Splendiani, A.; Sun, L.; Zhang, Y.; Li, T.; Kim, J.; Chim, C.-Y.; Galli, G.; Wang, F. Emerging photoluminescence in monolayer MoS₂. *Nano Lett.* **2010**, *10*, 1271–1275.

(5) Mak, K. F.; Shan, J. Photonics and optoelectronics of 2D semiconductor transition metal dichalcogenides. *Nat. Photonics* **2016**, *10*, 216.

(6) Podzorov, V.; Gershenson, M.; Kloc, C.; Zeis, R.; Bucher, E. High-mobility field-effect transistors based on transition metal dichalcogenides. *Appl. Phys. Lett.* **2004**, *84*, 3301–3303.

(7) Radisavljevic, B.; Radenovic, A.; Brivio, J.; Giacometti, V.; Kis, A. Single-layer MoS₂ transistors. *Nat. Nanotechnol.* **2011**, *6*, 147.

(8) Fang, H.; Chuang, S.; Chang, T. C.; Takei, K.; Takahashi, T.; Javey, A. High-performance single layered WSe₂ p-FETs with chemically doped contacts. *Nano Lett.* **2012**, *12*, 3788–3792.

(9) Gourmelon, E.; Lignier, O.; Hadouda, H.; Couturier, G.; Bernede, J.; Tedd, J.; Pouzet, J.; Salardenne, J. MS_2 (M= W, Mo) photosensitive thin films for solar cells. *Sol. Energy Mater. Sol. Cells* **1997**, *46*, 115–121.

(10) Lee, H. S.; Min, S.-W.; Chang, Y.-G.; Park, M. K.; Nam, T.; Kim, H.; Kim, J. H.; Ryu, S.; Im, S. MoS_2 nanosheet phototransistors with thickness-modulated optical energy gap. *Nano Lett.* **2012**, *12*, 3695–3700.

(11) Yin, Z.; Li, H.; Li, H.; Jiang, L.; Shi, Y.; Sun, Y.; Lu, G.; Zhang, Q.; Chen, X.; Zhang, H. Single-layer MoS_2 phototransistors. ACS Nano **2012**, 6, 74–80.

(12) Koppens, F.; Mueller, T.; Avouris, P.; Ferrari, A. C.; Vitiello, M.; Polini, M. Photodetectors based on graphene, other twodimensional materials and hybrid systems. *Nat. Nanotechnol.* **2014**, *9*, 780.

(13) Sun, Z.; Martinez, A.; Wang, F. Optical modulators with 2D layered materials. *Nat. Photonics* **2016**, *10*, 227.

(14) Carladous, A.; Coratger, R.; Ajustron, F.; Seine, G.; Péchou, R.; Beauvillain, J. Light emission from spectral analysis of Au/MoS_2 nanocontacts stimulated by scanning tunneling microscopy. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2002**, *66*, 045401.

(15) Sundaram, R.; Engel, M.; Lombardo, A.; Krupke, R.; Ferrari, A. C.; Avouris, P.; Steiner, M. Electroluminescence in single layer MoS₂. *Nano Lett.* **2013**, *13*, 1416–1421.

(16) Brumme, T.; Calandra, M.; Mauri, F. First-principles theory of field-effect doping in transition-metal dichalcogenides: Structural properties, electronic structure, Hall coefficient, and electrical conductivity. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2015**, *91*, 155436.

(17) Giannozzi, P.; Andreussi, O.; Brumme, T.; Bunau, O.; Buongiorno Nardelli, M.; Calandra, M.; Car, R.; Cavazzoni, C.; Ceresoli, D.; Cococcioni, M.; Colonna, N.; Carnimeo, I.; Dal Corso, A.; de Gironcoli, S.; Delugas, P.; DiStasio, R. A., Jr; Ferretti, A.; Floris, A.; Fratesi, G.; Fugallo, G.; Gebauer, R.; Gerstmann, U.; Giustino, F.; Gorni, T.; Jia, J.; Kawamura, M.; Ko, H.-Y.; Kokalj, A.; Küçükbenli, E.; Lazzeri, M.; Marsili, M.; Marzari, N.; Mauri, F.; Nguyen, N. L.; Nguyen, H.-V.; Otero-de-la-Roza, A.; Paulatto, L.; Poncé, S.; Rocca, D.; Sabatini, R.; Santra, B.; Schlipf, M.; Seitsonen, A. P.; Smogunov, A.; Timrov, I.; Thonhauser, T.; Umari, P.; Vast, N.; Wu, X.; Baroni, S. Advanced capabilities for materials modelling with quantum espresso. *J. Phys.: Condens. Matter* **2017**, *29*, 465901.

(18) Brumme, T.; Calandra, M.; Mauri, F. Electrochemical doping of few-layer ZrNCl from first principles: Electronic and structural properties in field-effect configuration. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2014**, *89*, 245406.

(19) Brumme, T.; Calandra, M.; Mauri, F. Determination of scattering time and of valley occupation in transition-metal dichalcogenides doped by field effect. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2016**, *93*, 081407.

(20) Cheiwchanchamnangij, T.; Lambrecht, W. R. Quasiparticle band structure calculation of monolayer, bilayer, and bulk MoS. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2012**, *85*, 205302.

(21) Ge, Y.; Liu, A. Y. Phonon-mediated superconductivity in electron-doped single-layer MoS_2 : a first-principles prediction. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2013**, *87*, 241408.

(22) Kadantsev, E. S.; Hawrylak, P. Electronic structure of a single MoS₂ monolayer. *Solid State Commun.* **2012**, *152*, 909–913.

(23) Fujimoto, T.; Awaga, K. Electric-double-layer field-effect transistors with ionic liquids. *Phys. Chem. Chem. Phys.* **2013**, *15*, 8983–9006.

(24) Ueno, K.; Shimotani, H.; Yuan, H.; Ye, J.; Kawasaki, M.; Iwasa, Y. Field-induced superconductivity in electric double layer transistors. *J. Phys. Soc. Jpn.* **2014**, *83*, 032001.

(25) Saito, Y.; Nojima, T.; Iwasa, Y. Gate-induced superconductivity in two-dimensional atomic crystals. *Supercond. Sci. Technol.* **2016**, *29*, 093001.

(26) Ye, J.; Zhang, Y.; Akashi, R.; Bahramy, M.; Arita, R.; Iwasa, Y. Superconducting dome in a gate-tuned band insulator. *Science* **2012**, 338, 1193–1196.

(27) Braga, D.; Gutierrez Lezama, I.; Berger, H.; Morpurgo, A. F. Quantitative determination of the band gap of WS_2 with ambipolar ionic liquid-gated transistors. *Nano Lett.* **2012**, *12*, 5218–5223.

(28) Yu, Y.; Yang, F.; Lu, X. F.; Yan, Y. J.; Cho, Y.-H.; Ma, L.; Niu, X.; Kim, S.; Son, Y.-W.; Feng, D.; Li, S.; Cheong, S.-W.; Chen, X. H.; Zhang, Y. Gate-tunable phase transitions in thin flakes of 1T-TaS₂. *Nat. Nanotechnol.* **2015**, *10*, 270.

(29) Xi, X.; Berger, H.; Forró, L.; Shan, J.; Mak, K. F. Gate Tuning of Electronic Phase Transitions in Two-Dimensional NbSe₂. *Phys. Rev. Lett.* **2016**, *117*, 106801.

(30) Chakraborty, B.; Bera, A.; Muthu, D.; Bhowmick, S.; Waghmare, U. V.; Sood, A. Symmetry-dependent phonon renormalization in monolayer MoS_2 transistor. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2012**, *85*, 161403.

(31) Das, A.; Pisana, S.; Chakraborty, B.; Piscanec, S.; Saha, S.; Waghmare, U.; Novoselov, K.; Krishnamurthy, H.; Geim, A.; Ferrari, A. C.; Sood, A. K. Monitoring dopants by Raman scattering in an electrochemically top-gated graphene transistor. *Nat. Nanotechnol.* **2008**, *3*, 210.

(32) Das, A.; Chakraborty, B.; Piscanec, S.; Pisana, S.; Sood, A.; Ferrari, A. C. Phonon renormalization in doped bilayer graphene. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2009**, *79*, 155417.

(33) Costanzo, D.; Jo, S.; Berger, H.; Morpurgo, A. F. Gate-induced superconductivity in atomically thin MoS₂ crystals. *Nat. Nanotechnol.* **2016**, *11*, 339.

(34) Lu, J.; Zheliuk, O.; Leermakers, I.; Yuan, N. F.; Zeitler, U.; Law, K. T.; Ye, J. Evidence for two-dimensional Ising superconductivity in gated MoS₂. *Science* **2015**, *350*, 1353–1357.

(35) Saito, Y.; Nakamura, Y.; Bahramy, M. S.; Kohama, Y.; Ye, J.; Kasahara, Y.; Nakagawa, Y.; Onga, M.; Tokunaga, M.; Nojima, T.; Yanase, Y.; Iwasa, Y. Superconductivity protected by spin-valley locking in ion-gated MoS₂. *Nat. Phys.* **2016**, *12*, 144–149.

(36) Chen, Q.; Lu, J.; Liang, L.; Zheliuk, O.; Ali, A.; Sheng, P.; Ye, J. Inducing and Manipulating Heteroelectronic States in a Single MoS₂Thin Flake. *Phys. Rev. Lett.* **2017**, *119*, 147002.

12955. (38) Roldán, R.; Cappelluti, E.; Guinea, F. Interactions and superconductivity in heavily doped MoS₂. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2013**, *88*, 054515.

valley electrons in transition metal disulfides. Nat. Commun. 2016, 7,

(39) Das, T.; Dolui, K. Superconducting dome in MoS_2 and $TiSe_2$ generated by quasiparticle-phonon coupling. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2015**, *91*, 094510.

(40) Kormányos, A.; Zólyomi, V.; Drummond, N. D.; Rakyta, P.; Burkard, G.; Fal'ko, V. I. Monolayer MoS_2 : trigonal warping, the Γ valley, and spin-orbit coupling effects. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2013**, *88*, 045416.

(41) Yuan, N. F.; Mak, K. F.; Law, K. Possible topological superconducting phases of MoS₂. *Phys. Rev. Lett.* **2014**, *113*, 097001.

(42) Costanzo, D.; Zhang, H.; Reddy, B. A.; Berger, H.; Morpurgo, A. F. Tunnelling spectroscopy of gate-induced superconductivity in MoS₂. *Nat. Nanotechnol.* **2018**, *13*, 483.

(43) Hsu, Y.-T.; Vaezi, A.; Fischer, M. H.; Kim, E.-A. Topological superconductivity in monolayer transition metal dichalcogenides. *Nat. Commun.* **2017**, *8*, 14985.

(44) Nakamura, Y.; Yanase, Y. Odd-parity superconductivity in bilayer transition metal dichalcogenides. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2017**, *96*, 054501.

(45) Lu, J. M.; Zheliuk, O.; Chen, Q.; Leermakers, I.; Hussey, N. E.; Zeitler, U.; Ye, J. T. Full superconducting dome of strong Ising protection in gated monolayer WS₂. *Proc. Natl. Acad. Sci. U. S. A.* **2018**, *115*, 3551.

(46) de la Barrera, S. C.; Sinko, M. R.; Gopalan, D. P.; Sivadas, N.; Seyler, K. L.; Watanabe, K.; Taniguchi, T.; Tsen, A. W.; Xu, X.; Xiao, D.; Hunt, B. M. Tuning Ising superconductivity with layer and spinorbit coupling in two-dimensional transition-metal dichalcogenides. *Nat. Commun.* **2018**, *9*, 1427.

(47) Cui, X.; Lee, G.-H.; Kim, Y. D.; Arefe, G.; Huang, P. Y.; Lee, C.-H.; Chenet, D. A.; Zhang, X.; Wang, L.; Ye, F.; Pizzocchero, F.; Jessen, B. S.; Watanabe, K.; Taniguchi, T.; Muller, D. A.; Low, T.; Kim, P.; Hone, J. Multi-terminal transport measurements of MoS_2 using a van der Waals heterostructure device platform. *Nat. Nanotechnol.* **2015**, *10*, 534.

(48) Kang, M.; Kim, B.; Ryu, S. H.; Jung, S. W.; Kim, J.; Moreschini, L.; Jozwiak, C.; Rotenberg, E.; Bostwick, A.; Kim, K. S. Universal mechanism of band-gap engineering in transition-metal dichalcogenides. *Nano Lett.* **2017**, *17*, 1610–1615.

(49) Pickett, W. E. Emergent Phenomena in Correlated Matter; Forschungszentrum Jülich GmbH and Institute for Advanced Simulations: Jülich, Germany, 2013.

(50) Fu, Y.; Liu, E.; Yuan, H.; Tang, P.; Lian, B.; Xu, G.; Zeng, J.; Chen, Z.; Wang, Y.; Zhou, W.; Xu, K.; Gao, A.; Pan, C.; Wang, M.; Wang, B.; Zhang, S.-C.; Hwang, H. Y.; Miao, F. Gated tuned superconductivity and phonon softening in monolayer and bilayer MoS₂. *npj Quantum Materials* **2017**, *2*, 52.

(51) Novoselov, K.; Jiang, D.; Schedin, F.; Booth, T.; Khotkevich, V.; Morozov, S.; Geim, A. Two-dimensional atomic crystals. *Proc. Natl. Acad. Sci. U. S. A.* **2005**, *102*, 10451–10453.

(52) Casiraghi, C.; Hartschuh, A.; Lidorikis, E.; Qian, H.; Harutyunyan, H.; Gokus, T.; Novoselov, K.; Ferrari, A. C. Rayleigh imaging of graphene and graphene layers. *Nano Lett.* **2007**, *7*, 2711– 2717.

(53) Kutz, M. Handbook of Materials Selection; John Wiley & Sons: Hoboken, NJ, 2002.

(54) Stacy, A.; Hodul, D. Raman spectra of IVB and VIB transition metal disulfides using laser energies near the absorption edges. *J. Phys. Chem. Solids* **1985**, *46*, 405–409.

(55) Verble, J.; Wieting, T. Lattice Mode Degeneracy in MoS_2 and Other Layer Compounds. *Phys. Rev. Lett.* **1970**, *25*, 362.

(56) Wieting, T.; Verble, J. Infrared and Raman Studies of Long-Wavelength Optical Phonons in Hexagonal MoS_2 . *Phys. Rev. B* **1971**, 3, 4286.

(57) Lee, C.; Yan, H.; Brus, L. E.; Heinz, T. F.; Hone, J.; Ryu, S. Anomalous lattice vibrations of single-and few-layer MoS_2 . ACS Nano **2010**, 4, 2695–2700.

(58) Zhang, X.; Han, W.; Wu, J.; Milana, S.; Lu, Y.; Li, Q.; Ferrari, A. C.; Tan, P. Raman spectroscopy of shear and layer breathing modes in multilayer MoS₂. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2013**, *87*, 115413.

(59) Tan, P.; Han, W.; Zhao, W.; Wu, Z.; Chang, K.; Wang, H.; Wang, Y.; Bonini, N.; Marzari, N.; Pugno, N.; Savini, G.; Lombardo, A.; Ferrari, A. C. The shear mode of multilayer graphene. *Nat. Mater.* **2012**, *11*, 294.

(60) Daghero, D.; Paolucci, F.; Sola, A.; Tortello, M.; Ummarino, G.; Agosto, M.; Gonnelli, R.; Nair, J. R.; Gerbaldi, C. Large conductance modulation of gold thin films by huge charge injection via electrochemical gating. *Phys. Rev. Lett.* **2012**, *108*, 066807.

(61) Zabrodskii, A. G.; Zinoveva, K. N. Low-temperature conductivity and metal-insulator transition in compensate n-Ge. *J. Exp. Theor. Phys.* **1984**, *59*, 425.

(62) Ren, Y.; Yuan, H.; Wu, X.; Chen, Z.; Iwasa, Y.; Cui, Y.; Hwang, H. Y.; Lai, K. Direct Imaging of Nanoscale Conductance Evolution in Ion-Gel-Gated Oxide Transistors. *Nano Lett.* **2015**, *15*, 4730.

(63) Shi, W.; Ye, J.; Zhang, Y.; Suzuki, R.; Yoshida, M.; Miyazaki, J.; Inoue, N.; Saito, Y.; Iwasa, Y. Superconductivity series in transition metal dichalcogenides by ionic gating. *Sci. Rep.* **2015**, *5*, 12534.

(64) El-Kareh, B. Fundamentals of Semiconductor Processing Technologies; Kluwer Academic Publishers: Dordrecht, The Netherlands, 1995.

(65) Ye, J.; Craciun, M. F.; Koshino, M.; Russo, S.; Inoue, S.; Yuan, H.; Shimotani, H.; Morpurgo, A. F.; Iwasa, Y. Accessing the transport properties of graphene and its multilayers at high carrier density. *Proc. Natl. Acad. Sci. U. S. A.* **2011**, *108*, 13002–13006.

(66) Gonnelli, R.; Piatti, E.; Sola, A.; Tortello, M.; Dolcini, F.; Galasso, S.; Nair, J. R.; Gerbaldi, C.; Cappelluti, E.; Bruna, M.; Ferrari, A. C. Weak localization in electric-double-layer gated few-layer graphene. 2D Mater. **2017**, *4*, 035006.

(67) Piatti, E.; Galasso, S.; Tortello, M.; Nair, J. R.; Gerbaldi, C.; Bruna, M.; Borini, S.; Daghero, D.; Gonnelli, R. Carrier mobility and scattering lifetime in electric double-layer gated few-layer graphene. *Appl. Surf. Sci.* **2017**, *395*, 37–41.

(68) Piatti, E.; Chen, Q.; Ye, J. Strong dopant dependence of electric transport in ion-gated MoS₂. *Appl. Phys. Lett.* **201**7, *111*, 013106.

(69) Gallagher, P.; Lee, M.; Petach, T. A.; Stanwyck, S. W.; Williams, J. R.; Watanabe, K.; Taniguchi, T.; Goldhaber-Gordon, D. A high-mobility electronic system at an electrolyte-gated oxide surface. *Nat. Commun.* **2015**, *6*, 6437.

(70) Ovchinnikov, D.; Gargiulo, F.; Allain, A.; Pasquier, D. J.; Dumcenco, D.; Ho, C.-H.; Yazyev, O. V.; Kis, A. Disorder engineering and conductivity dome in ReS_2 with electrolyte gating. *Nat. Commun.* **2016**, *7*, 12391.

(71) Lifshitz, I. Anomalies of electron characteristics of a metal in the high pressure region. *Sov. Phys. JETP* **1960**, *11*, 1130–1135.

(72) Gan, C. K.; Liu, Y. Y. F. Direct calculation of the linear thermal expansion coefficients of MoS_2 via symmetry-preserving deformations. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2016**, *94*, 134303.

(73) Standard Reference Material 739 Certificate; National Institute of Standards and Technology: Gaithersburg, MD, 1991.

(74) Nix, F.; MacNair, D. The thermal expansion of pure metals: copper, gold, aluminum, nickel, and iron. *Phys. Rev.* **1941**, *60*, 597.

(75) Yoon, D.; Son, Y.-W.; Cheong, H. Negative thermal expansion coefficient of graphene measured by Raman spectroscopy. *Nano Lett.* **2011**, *11*, 3227–3231.

(76) Mohiuddin, T.; Lombardo, A.; Nair, R.; Bonetti, A.; Savini, G.; Jalil, R.; Bonini, N.; Basko, D.; Galiotis, C.; Marzari, N.; Pugno, N.; Savini, G.; Lombardo, A.; Ferrari, A. C. Uniaxial strain in graphene by Raman spectroscopy: G peak splitting, Grüneisen parameters, and sample orientation. Phys. Rev. B: Condens. Matter Mater. Phys. 2009, 79, 205433.

(77) Klemens, P. Anharmonic decay of optical phonons. *Phys. Rev.* **1966**, *148*, 845.

(78) Lee, J.-U.; Woo, S.; Park, J.; Park, H.; Son, Y.-W.; Cheong, H. Strain-shear coupling in bilayer MoS₂. *Nat. Commun.* **2017**, *8*, 1370.

(79) Conley, H. J.; Wang, B.; Ziegler, J. I.; Haglund, R. F., Jr; Pantelides, S. T.; Bolotin, K. I. Bandgap engineering of strained monolayer and bilayer MoS₂. *Nano Lett.* **2013**, *13*, 3626–3630.

(80) Molina-Sanchez, A.; Wirtz, L. Phonons in single-layer and fewlayer MoS₂ and WS₂. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2011**, *84*, 155413.

(81) Ataca, C.; Topsakal, M.; Akturk, E.; Ciraci, S. A comparative study of lattice dynamics of three-and two-dimensional MoS₂. *J. Phys. Chem. C* **2011**, *115*, 16354–16361.

(82) Rösner, M.; Haas, S.; Wehling, T. Phase diagram of electrondoped dichalcogenides. *Phys. Rev. B: Condens. Matter Mater. Phys.* 2014, 90, 245105.

(83) Piatti, E.; Chen, Q.; Tortello, M.; Ye, J. T.; Gonnelli, R. S. Possible charge-density-wave signatures in the anomalous resistivity of Li-intercalated multilayer MoS₂. *Appl. Surf. Sci.* **2018**, DOI: 10.1016/j.apsusc.2018.05.232.

(84) Gor'kov, L. P.; Grüner, G. Charge Density Waves In Solids, 1st ed.; Elsevier: North Holland, 1989; Vol. 25.

(85) Gonnelli, R. S.; Daghero, D.; Tortello, M.; Ummarino, G. A.; Bukowski, Z.; Karpinski, J.; Reuvekamp, P. G.; Kremer, R. K.; Profeta, G.; Suzuki, K.; Kuroki, K. Fermi-Surface topological phase transition and horizontal Order-Parameter nodes in CaFe₂As₂ under pressure. *Sci. Rep.* **2016**, *6*, 26394.

(86) Allen, P. B.; Dynes, R. Transition temperature of strongcoupled superconductors reanalyzed. *Phys. Rev. B* 1975, *12*, 905.