Surface-bound chemical vapour deposition of carbon nanotubes: 
In situ study of catalyst activation

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Abstract

We combine in situ X-ray photoelectron spectroscopy and ex situ electron and Raman spectroscopy to study chemical interactions of SiO\textsubscript{2}-supported Fe catalyst films during C\textsubscript{2}H\textsubscript{2} exposure in the 400–600\,\textdegree{}C temperature range. Carbon nanotubes nucleate at C\textsubscript{2}H\textsubscript{2} pressures below 10^{-5} mbar, which allows a time-resolved recording of core level spectra. We find a rapid transition from an initial chemisorbed carbon on metallic Fe catalyst nanoparticles to a sp\textsuperscript{2} graphitic carbon network. Pre-treating the Fe catalyst film by H\textsubscript{2} sputtering influences the initial catalyst de-wetting and increases carbon nanotube yield.

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1. Introduction

Carbon nanotubes (CNTs) have extraordinary mechanical and thermal properties, like the highest Young’s modulus and highest axial thermal conductivity among any solid [1]. Moreover, single-wall nanotubes (SWNTs) have the highest current carrying capacity of any conductor, which makes them very attractive for use in future (hybrid) integrated circuits [2]. Since their discovery, much effort has been devoted to improve their quality and gain control on their characteristics by progressive development of the growth techniques. Chemical vapour deposition (CVD), being scalable and surface-bound, is one of the viable techniques to achieve full compatibility with the existing microelectronic processes [3,4]. The final aim would be to minimize the number of defects and to achieve a better control on properties such as diameter, length, and chirality. A complete knowledge of the catalyst substrate interaction, together with the understanding of the carbon precursor gas dissociation/de-hydrogenation mechanism during CVD, is fundamental to allow for a high degree of control and synthesize a specific tubular form of carbon.

Here, we present a systematic photoemission spectroscopy investigation of the chemical interaction between the catalyst and the carbon precursor gas, before and during CNT CVD, and discuss the role of the catalyst pre-treatment. Experiments are carried out in two different UHV systems. The catalyst deposition and the CVD steps are performed in sequence, without breaking the vacuum, to avoid any contamination and/or catalyst oxidation side effects. Monitoring of the Fe 2\textit{p}_{1/2}-2\textit{p}_{3/2} peaks allows us to assess the mechanisms involved during a H\textsubscript{2} sputtering pre-treatment performed on the catalyst film. The C 1s photoemission peak, during the carbon precursor gas...
dosing, is constantly monitored to reveal details about the
dynamics of the chemical changes involved during CNT
nucleation/growth.

2. Experimental

Two sets of X-ray photoemission spectroscopy (XPS)
experiments are performed. In situ Fe 2p photoemission
spectra are acquired in normal emission geometry, using a
conventional Mg X-ray source (\(hl=1253.6\) eV) with an
overall energy resolution of \(\sim 1.2\) eV. In situ time-resolved
XPS spectra from the C 1s core level are measured in the
UHV end station of the SuperESCA beamline (base
pressure \(<10^{-10}\) mbar) at the ELETTRA Synchrotron,
Trieste, Italy. A high flux of 400 eV photons allows core
level spectra to be acquired within 15 s, with an energy
resolution below 80 meV (probe size \(\sim 30 \times 200 \mu\)m²). We
stress that in situ, fast time-resolved XPS measurements
can only be carried out using a high-brilliance photon
source, such as a synchrotron light beam.

For all the experiments, we utilize commercial, polished
n-type Si(1 0 0) substrates, topped with a 150 nm thermally
grown SiO₂ film. Fe catalyst films are deposited in situ by
sublimation from heated filaments (Aldrich, 99.9% purity)
at a growth rate of \(\sim 0.6\) nm/h (calibrated by XPS).
Samples are clamped between two Ta contacts and heated
by direct heating. Fe films are always monitored by XPS to
detect the presence of any contaminant and/or silicides.
For CNT growth, undiluted C₂H₂ is let into the Super-
ESCA chamber using a gas-doser with a micro-channel
plate head, 15 mm away from the sample. This setup allows
in situ measurements in gas background pressures up to
5 \(\times 10^{-7}\) mbar. In the other chamber, C₂H₂ is let in through
a 1/4 in. stainless-steel pipe, whose nozzle is 1 cm away
from the sample. All samples are also characterized ex situ
by scanning electron microscopy (SEM, LEO 1530VP
FEGSEM) and Raman spectroscopy (Renishaw 1000
spectrometer, 633 nm laser excitation, 50 \(\times \) objective,
maximum power on the sample \(\sim 10^{7}\) W/m²).

We operate in the following parameter window:
\(\sim 0.08–0.6\) nm Fe film thickness, \(4 \times 10^{-8}–8 \times 10^{-3}\) mbar
C₂H₂ pressure and 400–600 °C CNT growth temperature.
A typical CVD route consists of a preliminary out-gassing
of the Si/SiO₂ substrate at 550–600 °C, followed by Fe
deposition at room temperature, and successive ramping-up
to the chosen CNT growth temperature. We also
investigate the effects of additional pre-treating of the
catalyst by H₂ sputtering (sputter gun, \(E_{\text{ion}} = 300\) eV) prior
to CNT growth.

3. Results

3.1. Role of catalyst pre-treatment

We detect an increase in CNT yield when pre-treating
the Fe catalyst film (performing a H₂ sputtering) prior to
growth. We detect SWNTs with the highest yield when
using \(\sim 0.15–0.6\)-nm-thick Fe films, sputtered with H₂ and
exposed at C₂H₂ pressures greater than \(8 \times 10^{-6}\) mbar, at
550–600 °C. Lower SWNT yields are generally observed at
lower pressures (<\(10^{-7}\) mbar). At even lower pressures,
and at temperatures below 550 °C, carbon nanofibres
(CNFs) or even no carbon structures are detected.

Fig. 1 shows the results of a CVD run with high SWNT
yield. Growth is performed by exposing an annealed
(at 580 °C, 5 min) and H₂ sputtered (at 580 °C, 5 min),
0.5-nm-thick Fe film (Si/SiO₂-supported) to undiluted
C₂H₂ (\(8 \times 10^{-3}\) mbar pressure) for 5 min at 580 °C. This

![Fig. 1](image)
confirms our previous observation that catalyst reconstruction by weak H$_2$ or NH$_3$ plasmas (or by purely thermal annealing in gaseous atmosphere) is beneficial to the increase of the CNT yield [5–8]. AFM analysis reveals that after such a pre-treatment, the catalyst reconstructs in smaller islands with a narrower diameter distribution, if compared to an annealing stage in vacuum, without pre-treatment [5–8]. The substrate used to grow the CNT sample in Fig. 1 was characterized by in situ XPS with the intent of monitoring any chemical changes of the Fe–SiO$_2$ interaction after each of the individual CVD steps prior to growth (catalyst deposition, annealing, H$_2$ sputtering). The evolution of the Fe 2$p$ spectra is shown in Fig. 1(b). The binding energy of the 2$p_{3/2}$ line of the as-deposited Fe (707.2 eV) corresponds to its metallic chemical state [9]. No peaks attributable to contaminants or Fe silicides are detected. After Fe annealing at 580 °C, no significant lineshape changes are observed. However, the absolute intensity of the Fe 2$p_{3/2}$ peak decreases by 35% (Fig. 1(b)). This is caused by the de-wetting of the Fe film on the SiO$_2$ substrate upon annealing [6,7,10], the islands diameter distribution depending on the initial film thickness and annealing temperature [11,12]. For our experiments, it is reasonable to assume that the Fe film is characterized by larger islands before being subjected to the H$_2$ sputtering [4,7,8,13]. A confirmation of the “morphological” effect of the H$_2$ sputtering is suggested by the analysis of the Fe 2p photoemission spectrum (Fig. 1(b)). We find the Fe 2$p$ peak intensity to increase after H$_2$ sputtering with respect to the corresponding Fe 2$p$ signal measured after UHV annealing. This increase indicates that Fe covers more substrate surface after H$_2$ sputtering than after an annealing stage in UHV, implying that islands diameters and heights are larger after UHV annealing than after H$_2$ sputtering. No Fe 2$p$ lineshape changes are observed throughout all the steps of catalyst preparation, as also confirmed by UV photoemission measurements. We can therefore rule out any chemical effect of H$_2$ sputtering on the catalyst. H$_2$ sputtering could create Fe–SiO$_2$ bonds [14], preventing Fe sintering during sample heating. Partial burial of Fe energy of the 2$p$ of strong Fe–SiO$_2$ bonds can be ruled out. We cannot indicate that Fe remains metallic. Therefore, the formation of SWNT nucleation.

3.2. Chemical changes during CNT nucleation

The Fe–C$_2$H$_2$ interaction during the CNT growth stage is investigated in the SuperESCA chamber through in situ monitoring of the C 1$s$ peak time evolution [20]. No C 1$s$ peak is detected on samples where the Fe catalyst is not present, thus ruling out any possible interaction between C$_2$H$_2$ and the SiO$_2$ surface in our experimental conditions.

Fig. 2 shows the results of CNT CVD performed on to H$_2$ sputtered Fe films in the same experimental conditions as for Fig. 1, but at lower C$_2$H$_2$ pressure (2 × 10$^{-7}$ mbar). Fig. 2(b) shows the time-resolved evolution of the C 1$s$ peak. Its dynamics can be described as follows: as soon as C$_2$H$_2$ is let into the chamber, a peak at 282.6 eV appears, indicating that C is chemisorbed on the Fe catalyst [21,22]. After 90 s incubation, another peak at 283.2 eV appears, persisting for 30 s. We attribute this peak to the formation of carbidic C [21,23]. The formation of a sp$^2$N graphitic C network is detected by the appearance of another peak at 284.5 eV (corresponding to the typical C 1$s$ binding energy of the C–C bond) [21,24]. The intensity of the latter peak increases very rapidly: after 15 s, the chemisorbed C and carbidic C peaks drastically reduce. The graphitic peak saturates after 150 s, suggesting a termination of CNT growth, and dominates over the chemisorbed C and carbidic C peaks. A SEM picture of the as-grown sample is presented in Fig. 2(a), showing the presence of laterally extended, sparse CNTs.

Raman spectroscopy is a fast and non-destructive method for characterization of carbon materials [25]. In the case of SWNTs, important information such as diameter, orientation, metallic or semiconducting character and chirality, can be obtained from a Raman spectrum [26–30]. Fig. 3(a) shows a Raman spectrum (633 nm) of the sample shown in Fig. 2(a). We observe a structured G peak at ~1600 cm$^{-1}$, typical of nanotubes [26–30], and the D peak at ~1300 cm$^{-1}$, indicative of defects and disordered graphitic material [31]. The presence of well-defined radial breathing modes (RBMs) in the low frequency range (inset in Fig. 3), together with the shape of the G peak, is the typical signature of SWNTs [26,27]. SWNT diameters (corresponding to the RBMs observed) can be calculated using $d = C_1/(\omega_{RBM} - C_2)$, with $C_1 = 214.4$ nm/cm and $C_2 = 18.7$ cm$^{-1}$ [28], resulting in a 0.8–1.3 nm diameter distribution. Due to the cut-off of our notch filter, we cannot detect SWNT diameters >2 nm.

The C 1$s$ peak time evolution (Fig. 2(b)) shows an increase of binding energy throughout the continuous transition from a chemisorbed C form (282.6 eV) [21,22] to carbidic C (283.2 eV) [21,23], and ultimately to a
re-organization of the C atoms in a $sp^2$ network (284.5 eV, CNT growth) [21,24]. Our data indicate that the chemisorbed C state persists up to 25 at% C concentration. At higher concentrations, the C–Fe bond changes, and the carbidic C signature at 283.2 eV appears.

In literature, a binding energy of 282.6 eV, measured by XPS on the surface of Fe single crystals annealed at high temperatures, was attributed to chemisorbed C [21,22]. In Ref. [22], chemisorbed C of sub-monolayer thickness was measured after sample heating up to temperatures in the 377–577 °C range. In Ref. [21], the Fe sample was heated at temperatures between 450 and 700 °C, obtaining again sub-monolayer-thick chemisorbed C or graphitic C on the metal surface as separate phases, reversibly transformed into...
each other under specific annealing conditions. We thus use Ref. [21] to define the reference binding energies for chemisorbed C and graphitic C, since they characterized by XPS a system similar to ours (carbon bonded to Fe).

We observe a similar time evolution of the C 1s peak when using slightly different CVD conditions for CNT growth. Fig. 3(b) shows the Raman spectrum (633 nm) of sample grown by exposing a H2 sputtered (5 min), 0.4-nm-thick Fe film to undiluted C2H2 (4 × 10⁻8 mbar pressure) for 5 min at 600 °C. The spectrum is characterized by very broad D and G peaks, typical of disordered carbons and non-highly crystalline tubular structures such as CNFs [31]. However, in situ monitoring of the C 1s peak time evolution (in the SuperESCA chamber) reveals a dynamics very similar to that described for the SWNT sample in Fig. 2. We attribute the differences between the samples whose Raman is in Fig. 3(a) and (b) to the concurrently effects of the changes in H2 sputtering time duration prior to growth and to the different C2H2 pressure.

We are currently investigating the effects of hydrocarbon gas dosing, and preliminary results show that the tendency to form defective tubular carbon structures depends on the rate of carbon feeding: fast hydrocarbon gas pressure gradients can result in CNT growth at low pressures (2 × 10⁻⁷ mbar), while higher pressures are less sensitive to such gradients [32].

4. Conclusions

We presented an investigation of CNT growth by CVD in UHV conditions, by exposing in situ deposited Fe films on Si/SiO2 to undiluted C2H2 at pressures in the 4 × 10⁻⁸–8 × 10⁻³ mbar range. Continuous monitoring of the relevant photoemission peaks allow us to conclude that a catalyst pre-treatment (H2 sputtering) during CNT growth can significantly improve the CNT yield. No Fe 2p lineshape changes were observed throughout all the steps of catalyst preparation, hence we rule out the formation of strong Fe–SiO2 bonds. Chemical changes of the Fe–C2H2 interaction occurring in the early stages of CNT nucleation/growth were monitored by time-resolved photoemission. We observed a rapid transition from an initial chemisorbed C on the metallic Fe catalyst to a sp² C network occurs, through the formation of an intermediate carbidic stage.

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