# Photonics with Multiwall Carbon Nanotube Arrays

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**ABSTRACT** We investigate the photonic properties of two-dimensional nanotube arrays for photon energies up to 40 eV and unveil the physics of two distinct applications: deep-UV photonic crystals and total visible absorbers. We find three main regimes: for small intertube spacing of 20–30 nm, we obtain strong Bragg scattering and photonic band gaps in the deep-UV range of 25~35 eV. For intermediate spacing of 40–100 nm, the photonic bands anticross with the graphite plasmon bands resulting into a complex photonic structure, and a generally reduced Bragg scattering. For large spacing >150 nm, the Bragg gap moves into the visible and decreases due to absorption. This leads to nanotube arrays behaving as total optical absorbers. Our results can guide the design of photonic applications in the visible and deep UV ranges.

**KEYWORDS:** carbon nanotube  $\cdot$  nanotube array  $\cdot$  photonic crystal  $\cdot$  effective medium theory  $\cdot$  deep UV photonics  $\cdot$  ultrablack materials

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arbon nanotubes (CNTs) are at the center of nanotechology research. The search for new applications based on their mechanical and electrical properties is ever increasing.<sup>1</sup> Examples include high-strength reinforced materials, tips in atomic force microscopes, field emission devices, field effect transistors, electrical contacts and interconnects, as well as nanoelectrodes for liquid-crystal photonic devices, and many others.<sup>1</sup> It is only natural to think that a similar richness in properties, and thus in applications, could be expected for the photonic response of these materials. CNTs are strongly anisotropic<sup>2–15</sup> and highly nonlinear.<sup>15–21</sup> They are ideal saturable absorbers for applications in ultrafast lasers,<sup>22–30</sup> total optical absorbers for extremely dark materials<sup>31</sup> or nanoantennas for optical frequencies.<sup>33,34</sup> They are good infrared (IR) absorbers that can function as IR bolometric detectors,<sup>35</sup> as well as good IR emitters.<sup>36,37</sup> It was also suggested that regular arrays of CNTs could serve as photonic crystals.38-43

Photonic crystals<sup>44–47</sup> are structures with a spatial periodicity in their dielectric constant, giving rise to strong multiple scattering of the impending electromagnetic

waves when their period is comparable to the incident wavelength (Bragg scattering). If the scattering is strong enough, complete spectral gaps may open where light propagation is prohibited in all directions. These are called photonic band gaps. Many important effects and applications of photonic crystals, such as light localization,48 inhibition of spontaneous emission,49 light guidance and manipulation,<sup>50,51</sup> and, more recently, superlensing<sup>52</sup> and meta-materials exhibiting negative refraction,<sup>53,54</sup> have been extensively studied both theoretically and experimentally. The materials typically used up to date have been semiconductors,<sup>51,55,56</sup> polymers,<sup>57-59</sup> and metals.<sup>60–63</sup> In terms of CNT-based photonic crystals, however, little has been done. In particular, some evidence of Bragg diffraction has been reported,<sup>39-41</sup> and an effective medium theory has been applied in the long wavelength limit.<sup>42,43</sup> To the best of our knowledge no measurements or calculations have yet been performed proving the existence and properties of full photonic band gaps in periodic arrays of aligned CNTs. Here, we present full photonic band structure and reflectivity calculations and explore the CNT performance and potential applications in the technologically important area of photonic crystals.

CNTs consisting of a single rolled-up graphene layer are called single-wall carbon nanotubes (SWNTs), while multiwall carbon nanotubes (MWNT) have two or more walls. It is now possible to grow arrays of CNTs in defined positions, orientations and lengths, down to low temperatures.<sup>64–70</sup> While SWNTs have diameters typically on the order of a nanometer, MWNTs can be controllably deposited and aligned with diameters up to hundreds nanometers.<sup>64–68</sup> Bragg scattering

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occurs when the periodicity is comparable to the wavelength of light. For a given periodicity, this is maximized when the scatterer size is comparable to the spacing, that is, for volume filling ratios between 5% and 50%,<sup>44</sup> or, diameter/spacing ratios between 0.25 and 0.8. Given the large range of achievable MWNT diameters, it is in principle expected that MWNT arrays may induce strong Bragg-like scattering for a wide frequency range between visible and UV, and thus may be ideal candidates for Vis–UV photonics.

An important feature, which defines the photonic properties of CNT arrays, is their dielectric response. This is anisotropic.<sup>2–15</sup> However, apart from the cylindrical geometry, the general features observed are similar to those of bulk graphite.<sup>10–12,15</sup> Given that CNTs are rolled-up graphene layers, this is somewhat expected: graphite is highly anisotropic, with different dielectric response functions  $\varepsilon_{\parallel}(\omega)$  and  $\varepsilon_{\perp}(\omega)$  depending on the electric field being polarized along or perpendicular to its *c*-axis. For a CNT, if the electric field is polarized parallel to its axis, it is also parallel to the graphite layers, otherwise both parallel and perpendicular contributions exist. Graphene, that is, a single graphite layer, has been shown to have a similar optical response to bulk graphite for light incident normal to its plane and polarized along it.71-74 Assuming curvature effects to be small and the local dielectric function to be well described by that of graphite, for electric field polarized parallel to the CNT axis we have only the  $\varepsilon_{\perp}(\omega)$  contribution, while for perpendicular we need consider both  $\epsilon_{\perp}(\omega)$  and  $\epsilon_{\parallel}(\omega)$ . Such "rolled-up" graphite was used to model CNTs electron energy loss spectra (EELS),75-78 and good agreement was found down to the SWNT scale. A "spherical" graphite model was also invoked to explain EELS of multishell fullerenes.<sup>77-82</sup> We thus assume bulk graphite to be adequate for the dielectric function of MWNTs with more than 10 walls (diameters larger than 7 nm). Indeed, graphene flakes with >10 layers are indistinguishable from graphite.71,83,84

# **RESULTS AND DISCUSSION**

We study the photonic response of two-dimensional MWNT square arrays under transverse magnetic (TM) polarized illumination (i.e., MWNT infinitely long, with light polarized parallel to the CNT axis, and incident along a normal plane). In reality, of course, MWNT are not infinite, but, as long as their length is several times their diameter and the incident light wavelength, this approximation is valid. Since in photonic band gap applications the scatterer diameter must be comparable to the wavelength of light, the length/diameter ratios of more than 50, which can be routinely produced in MWNT arrays, definitely satisfy these requirements.<sup>64–66</sup> TM is the optimal polarization for opening a photonic band gap in an array of dielectric cylindrical objects in vacuum.<sup>44</sup> For this polarization, both square and hexagonal arrays are good candidates



Figure 1. (a) Schematic of geometry: two dimensional square array of MWNTs with diameter *d* and separation *a*. (b) Two dimensional projection of the system. We study the photonic properties in the TM polarization, *i.e.*, for light polarized with electric field parallel to the tube axis. (c) Reciprocal lattice of structure in panel b: light blue shading is the first Brillouin zone, and dark blue shading is its irreducible part.

and yield similar qualitative results.<sup>44</sup> Given its simplicity in setup and analysis, we here focus on the square array. Furthermore, for this polarization only  $\varepsilon_{\perp}(\omega)$  is needed, that is, that best measured experimentally.85 Finally, we assume nanotubes with interior walls all the way to their center. A schematic of this arrangement and its 2D projection are shown in Figures 1a,b, respectively. The spatial periodicity of the MWNT array results into an equivalent periodicity in the reciprocal space, with a reciprocal lattice unit vector of magnitude |G| = $2\pi/a$ , with *a* the inter-MWNT lattice spacing, Figure 1c. The reciprocal lattice primitive cell is the first Brillouin zone, shown in light blue shading in Figure 1c, while in dark blue shading is the irreducible part of the first Brillouin zone, revealing the independent photon wavevectors and their symmetry. The most important,



Figure 2. Dielectric function of graphite for the  $E \perp c$  polarization. Solid lines represent the model used in this work, while symbols indicate the model used in ref 85.

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Figure 3. Computational cell for (a) transmission and reflection calculation; periodic boundary conditions on the side, perfectly matched-layer (absorbing) boundary conditions at the ends. A wide-spectrum plane source is used. (b) Band structure calculation; periodic boundary conditions all around the unit cell. A wide spectrum point-source and a monitor point are used, both on nonsymmetric points of the unit cell. In both cases the computational discretization is 1 nm/ grid.

in terms of band structure and gap formation, are the ones tracing its periphery along  $\Gamma \rightarrow X \rightarrow M \rightarrow \Gamma$ .<sup>44</sup>

We use the graphite in-plane dielectric function  $\epsilon_{\perp}(\omega)$  proposed by ref 85 (see also refs 86–88), which is a compilation of data from several groups, consistent with continuity and Kramers–Kronig requirements. This can be fitted with a Drude–Lorentz model:<sup>89</sup>

$$\varepsilon_{\perp}(\omega) = 1 - \frac{\omega_{p}^{2}}{\omega^{2} + i\omega/\tau} + \sum_{m=1}^{M} \frac{\sigma_{m}^{2}}{\omega_{m}^{2} - \omega^{2} - i\omega\gamma_{m}} (1)$$

where the first term is the Drude free electron contribution, the second contains Lorentz oscillators corresponding to interband transitions.  $\omega_p$  and  $\tau$  are free electron plasma frequency and relaxation time,  $\hbar\omega_m$ ,  $\sigma_m$ , and  $\gamma_m$  are transition energy, oscillator strength, and decay rate for the Lorentz terms. To accurately reproduce the experimental graphite dielectric function we treat these as fit parameters. Reference 85 reported an excellent fit to the experimental data using M = 7. However, they also had frequency-dependent decay rates

# TABLE 1. Parameters Used for Our Graphite DielectricFunction Fit Shown in Figure $2^a$

т	$\gamma_m$	ω_m	σm
0	2.32	0.00	2.09
1	1.58	0.74	4.32
2	2.59	2.38	5.19
3	1.60	4.46	8.25
4	1.01	14.42	3.61
5	2.17	14.12	14.63
6	9.94	16.36	14.97
7	8.40	21.45	6.05

<sup>*a*</sup>All numbers are in eV. The Drude term is for m= 0, where  $\gamma_0\equiv$  1/au and  $\sigma_0\equiv$ 

 $\gamma_m(\omega)$ , which is incompatible with our numerical method. Thus, we perform a new fit with static decay rates, using again M = 7. Our model parameters are shown in Table 1, while Figure 2 plots our model dielectric function  $\varepsilon_{\perp}(\omega)$  and that of ref 85, showing that they are almost identical.

The important features to note in the  $\varepsilon_{\perp}$  of the graphite are, in the imaginary part, the two absorption peaks due to the  $\pi \rightarrow \pi^*$  transition at ~4.5 eV and the  $\sigma \rightarrow \sigma^*$  at ~14.3 eV. In Re{ $\varepsilon_{\perp}$ } these are accompanied by metallic bands (i.e., energy regions with negative Re{ $\varepsilon_{\perp}$ }<sup>89</sup>) occurring just above those transitions, indicated in Figure 2 by  $M_{\pi}$ ,  $M_{\sigma}$ , which terminate at plasma energies of ~7 and 27 eV, respectively<sup>85</sup> (where Re{ $\varepsilon_{\perp}$ } = 0). For low energies both real and imaginary part increase, indicating large absorption, while at high energies the response asymptotically approaches zero with Re{ $\varepsilon_{\perp}$ }  $\rightarrow$  1 and Im{ $\varepsilon_{\perp}$ }  $\rightarrow$  0.

 We use the finite-difference time-domain (FDTD) method<sup>90-92</sup> to calculate (1) reflectivity and effective medium properties, and (2) photonic band structure. For the former we solve the transmission problem of a single array row, as shown in Figure 3a (where the CNT is periodically repeated in the vertical direction because of the periodic boundary condition). For the photonic band structure, the computational cell consists of a single unit cell, as shown in Figure 3b, where the CNT is periodically repeated in both vertical and horizontal directions because of the periodic boundary conditions (*i.e.*, we simulate a "bulk" 2D array). More details on these calculations are given in the Methods section.

The onset of multiple scattering effects is determined by the value of a. This happens for  $\lambda \approx 2a\langle n \rangle$ . We first calculate the response of a square array with a = 20 nm and surface filling ration f = 35% (i.e., diameter d = 13.5 nm). At this spacing, Bragg scattering is expected around  $\lambda \approx 35$  nm (~35 eV). This puts us close to the edge of the dielectric-function model range (see Figure 2) but fairly far from the two plasmons, so absorption will be minimal and thus there will be higher chances to observe multiple scattering. The chosen *f* is typical to promote Bragg scattering effects. If the values are too large absorption dominates (for f > 50%), while if too small the scattering cross sections diminishes (for f < 5%).

Figure 4 plots the calculated effective material constants for this system. In the long wavelength limit  $\lambda \gg a$  (e.g.,  $\lambda > 60$  nm or  $\hbar \omega < 20$  eV), the effective permittivity is very similar to that of bulk graphite up to some scaling factor. This means that no resonant effects develop and the effective response is merely a volume average of the dielectric function of the individual constituents,<sup>42</sup> that is, MWNT and vacuum. At the Bragg resonance,  $\hbar \omega \approx 33$  eV (or  $\lambda \approx 37$  nm), on the other hand, a strong effective permittivity and mag-

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 $\omega_{\rm p}$ 





Figure 4. Effective medium parameters for the a = 20 nm, f = 35% square array of MWNTs.

netic response develop. These are characteristic of photonic crystals inside the first band gap,<sup>93</sup> that is, how an effective uniform medium incorporates multiple reflections occurring in the discrete system at the Bragg condition. This is a signature of strong Bragg-like multiple scattering in deep UV.

With these effective material parameters we calculate the expected response of any number of MWNT array rows. Figure 5c-e plot reflection, absorption, and transmission from 10 rows at a = 20 nm and f = 35%. This amounts to a total system length of 200 nm, Figure 5a. For comparison we also plot the response of a solid graphite film, with thickness chosen to have the same density per unit cross section, that is, 200 nm ×

0.35 = 70 nm, Figure 5b. This ensures that light will effectively "cover" the same amount of graphitic material for both systems, thus serving as a "normalized" comparison between the two responses.

The major difference is at the Bragg reflectivity peak, which is totally absent in the bulk case. This is accompanied by a strong dip in the absorption and transmission. On the other hand, both systems have metallic peaks in the reflectivity and dips in absorption ( $M_{\pi}$  and  $M_{\sigma}$  in Figure 5c-e), with the bulk graphite response being more pronounced. This is expected since for these energies the array behaves as bulk, being in the subwavelength limit  $\lambda \gg a$ . Indeed, Figure 4a shows that, while the effective dielectric constant  $\tilde{\epsilon}$  is still metallic (i.e., negative), it is however less pronounced, thus allowing more light to penetrate. This gives less reflection and more absorption. In general, far from the Bragg peak, the array is a better absorber than a reflector. However the transmission is much more suppressed for the array system, close to 2 orders of magnitude at the  $\pi \rightarrow \pi^*$  transition energy and 5 orders of magnitude at the  $\sigma \rightarrow \sigma^*$  transition energy. The array system, thus, promotes stronger absorption.

To explore the structural parameter space, we vary a, with f = 35%. Figure 6a plots the semi-infinite reflection curves, each shifted by 0.25 for clarity. As a increases, the Bragg peak shifts to lower energies, and an-



Figure 5. (a) Scheme of a 10-row MWNT system; (b) scheme of a 70nm bulk graphite system; (c–e) response of 10 rows MWNT array with  $\alpha = 20$  nm and f = 35%. The dotted line is the response of 70 nm thick graphite: (c) reflection, (d) absorption, and (e) transmission.



Figure 6. Semi-infinite reflection curves for various lattice constants and (a) f = 35%; (b) f = 10%. Each terminates at the diffraction limit  $\lambda = a$ . Shaded vertical bars mark the two metallic bands, whose position is independent from the spacing. Orange arrows point out the Bragg peaks, while brown arrows point out the new Bragg peaks appearing below the metallic bands due to anticrossing.

ticrosses the  $M_{\sigma}$  metallic band. At a = 30 nm this causes a new peak to appear, just below  $M_{\sigma}$ , and at a = 40nm the new peak is much more pronounced, while the original Bragg peak merges with  $M_{\sigma}$ . One way to understand this is that, as the Bragg condition shifts to lower energies, the dielectric constant changes and, thus, the condition self-consistently readjusts to a new energy. In some cases the Bragg condition gets partially destroyed and resatisfied at a lower energy, appearing as if the Bragg peak splits in two. At even larger *a*, the Bragg peak moves between  $M_{\pi}$  and  $M_{\sigma}$ , but is less pronounced because of increased absorption. Above a =100 nm, it crosses  $M_{\pi}$  and moves into the visible. There, it gets smeared out, because of absorption. Note that for all a,  $M_{\pi}$  and  $M_{\sigma}$  remain at the same energy.

The response evolution is not so pronounced if we assume a smaller f = 10%, Figure 6b (for the smallest



a = 20 nm this implies a diameter d = 7 nm). We observe a narrower and taller Bragg peak and, at the same time, narrower but shorter metallic peaks. As the Bragg peak shifts to lower energies for increasing separations, it does not significantly anticross the metallic bands. In particular, only for a = 40 nm do we see a small repulsion between the two peaks, still not strong enough to produce the splitting observed for f = 35%. This is expected because both bands are very narrow, with a much smaller overlap. In most cases this gives much smaller interaction, except for a = 40 nm, where both bands have the same energy.

To elucidate the Bragg- $M_{\sigma}$  anticrossing for the f =35% case, we calculate the band structure for a = 20and 40 nm in Figures 7b and 8b. These also plot the corresponding normal incidence semi-infinite reflection, for direct comparison. The red bands mark the metallic gaps, orange marks the partial Bragg gaps, while the yellow band in Figure 8b marks a full Bragg gap. We associate every peak in the reflectivity with the gaps in the band structure. For a = 20 nm, we only have a partial Bragg gap, and all three gaps (Bragg,  $M_{\alpha}$ , and  $M_{\pi}$ ) are well separated. The photonic band structure around the metallic gaps results from the anticrossing of the free photon band and the localized tube modes below the  $\pi \rightarrow \pi^*$  or  $\sigma \rightarrow \sigma^*$  transitions (where the dielectric constant has large positive values), and is similar to what observed in 2D arrays of polaritonic materials in the THz regime.<sup>92</sup> For a = 40 nm, the band structure is more complex. The Bragg gap evolves into a full band gap, while a new propagating mode appears below  $M_{\alpha}$ (where, as discussed above,  $Re\{\epsilon_{\perp}\}$  has large positive values), resulting into a new partial Bragg gap. This complex behavior resulting from the band anticrossing could be combined with measurements to become a valuable diagnostic tool of dielectric function and geometric structure of MWNTs arrays.

The photonic eigenmodes at high-symmetry points are shown as insets in Figures 7b, and 8b. In Figure 7b we separate two different regions: below 30 eV the modes are localized resonances supported by the high dielectric constant (i.e., below the  $\pi \rightarrow \pi^*$  and  $\sigma \rightarrow \sigma^*$ transitions). Above 30 eV the modes resemble the free photon bands with a weak scattering perturbation. The only difference from what usually is observed in 2D photonic crystals<sup>44</sup> is that, since here  $\text{Re}\{\epsilon_{\perp}\} < 1$  (see Figure 2), the order in increasing energy in which the modes appear is reversed, that is the s-type monopole mode appears at higher energy than the *p*-type dipole. In Figure 8b the free photon bands move downward and anticross the localized modes, resulting into a new branch and partial Bragg gap at  $\sim$ 12 eV. The modes in either side of this gap are similar, but in energy reverse order from those at the top gap of 20 eV, since here  $Re\{\epsilon_{\perp}\} > 1$ . These modal patterns may also be useful to study surface and impurity states in MWNT arrays.<sup>44</sup>

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Figure 6 indicates that, apart from the inter-MWNT spacing, the photonic response is also dependent on the surface-filling ratio. Given that this is a very important experimental parameter, we explore it more in Figure 9a, where the reflection from 5 array rows at fixed a = 20 nm is plotted for seven different f. The main trend is, as expected, that reflection increases for increasing f: For fixed spacing, increasing f amounts to increasing the MWNT diameter and, thus, the scattering cross section. There is a difference, however, between the two metallic peaks and the Bragg peak. While the metallic peak position does not shift in energy and monotonously increases in magnitude, the Bragg peak blue-shifts and reaches a maximum at  $f \approx 35\%$ , after which it decreases. The blue-shift is explained from the fact that, since the Bragg peak appears at  $\lambda \approx 2a\langle n \rangle$ , and given that at these energies the tube refractive index is less than 1 (because  $\text{Re}\{\epsilon_{\perp}\} < 1$ ), increasing diameter decreases the average index and thus blue-shifts the response. The nonmonotonic response at large f is because absorption increases due to increased material coverage, and thus, Bragg scattering reduces. At even larger f, scatterers start to overlap, further reducing the response, which eventually equates the bulk graphite reflectivity at the  $f \rightarrow 100\%$  limit.

It is even more interesting to see how the response is modified when we remove the finite thickness restriction, that is, look at the semi-infinite reflection for different filling ratios, as plotted in Figure 9b. The metallic peak behavior of the semi-infinite system is hardly different from before, from which we conclude that the system's response already saturates at five rows. Thus, at these energies the system behaves effectively as a metal with a small penetration depth. In contrast, the Bragg peak has maximum reflectivity at the smallest  $f_{i}$ with the peak magnitude monotonically decreasing for increasing f. The blue-shift is the same as in the 5-row case, but in addition we notice a strong decrease in bandwidth for the smaller f. The reason is that the structure is now semi-infinite and, thus, there are enough array rows for the Bragg backscattering mechanism to build-up, even for the smallest f. In this case the only remaining limiting factor is absorption: the smaller *f* has less absorption and, thus, the maximum peak reflection, reaching above 70% as f goes to zero. This, however, is only true for energies exactly inside the corresponding Bragg gap. Since this decreases with f, so does the reflection peak bandwidth.

Thus, the optimal surface filling ratio  $f_{max}$  is different depending on the number of array rows. Figure 10a plots the calculated peak reflection for a = 20 nm for different number of rows as a function of *f*. Indeed, depending on the number of rows, the peak reflection reaches a specific maximum as a function of *f*. This moves to smaller  $f_{max}$  for increasing array rows. This trend is shown in Figure 10b, which plots the maximum reflection and tube diameter corresponding to



Figure 8. (a) Semi-infinite reflectivity and (b) photonic band structure for a = 40 nm; f = 35%; d = 27 nm.

 $f_{\rm max}$  as a function of the number of array rows. Thus, we can tailor the photonic response of MWNT arrays in the deep UV range. These results may find application in emerging UV technologies such as extreme UV lithography.<sup>94–97</sup>

Thus far we focused on the UV spectrum, i.e. *a* small enough so that the Bragg peak is far from the metallic bands. This is because, as shown in Figure 6, for increasing *a* the Bragg peak intermixes with the metallic bands and diminishes in strength. We now turn to the other side of the spectrum, at *a* large enough so that the Bragg peak appears at a much lower energy than the  $\pi \rightarrow \pi^*$  peak, that is, optical frequencies. This is the region where most proposals for CNT-based photonic crystals have been reported.<sup>38–41</sup> The dielectric function there (inset of Figure 11) is characterized by very large absorption: a moderately large real part  $\varepsilon_1 \equiv$ 



Figure 9. Normal reflection for a = 20 nm and varying f: (a) five array rows, (b) semi-infinite.

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Figure 10. (a) Peak Bragg reflection for the a = 20 nm array as a function of f for five systems with different number of array rows. (b) Optimal diameter and maximum achieved Bragg reflection as a function of the number of rows used for the a = 20 nm array.

 $Re\{\epsilon_{\perp}\}$  accompanied by a much larger imaginary part  $\varepsilon_2 \equiv Im\{\varepsilon_{\perp}\}$ . Figure 11a shows the semi-infinite normalincidence reflection when a = 500 nm, d = 113 nm, and f = 4%. These parameters are very close to those proposed by refs 38-41 and typical for aligned MWNT arrays.<sup>64–66</sup> We do observe Bragg peaks, the fundamental reaching up to 50%. The higher order Bragg peaks are smaller and narrower, as usually expected for photonic crystals.<sup>44</sup> While not significantly lower than Figure 9 for a = 20 nm and f = 5%, this is not particularly encouraging, given the variety of semiconducting materials and versatile processing technology that can yield photonic crystals at these frequencies.55,56 Thus, we do not expect pristine MWNT arrays to play a competitive role in applications requiring visible light reflectors and/or modulators. As pointed out above, this is due to the large absorption of graphite at optical frequencies, which limits Bragg backscattering. On the other hand, coated MWNTs 98-100 or MWNTs encapsulating nanowires,<sup>101</sup> having a different dielectric func-



Figure 11. (a) Semi-infinite reflectivity and (b) photonic band structure for a = 500 nm; f = 4%; d = 113 nm. In the inset, the in-plane graphite dielectric function for visible photon energies is shown. The band structure in b is calculated neglecting absorption.

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tion, could play a significant role in this frequency region.

For completeness, Figure 11b also plots the corresponding band structure of a nonabsorbing material with  $\epsilon = 5.25$  (due to its extreme value, it is not possible to obtain a meaningful photonic band structure including the absorption). The peaks in reflection match well with the partial gaps in the band structure. The dashed red line in Figure 11b marks the diffraction limit  $a = \lambda$ .

We further explore the MWNT array response by calculating the reflection and transmission at different  $f_i$ keeping a = 500 nm. Figure 12a plots the total scattering (reflection + transmission) for 10 array rows (i.e., a film thickness of 5  $\mu$ m). We note that very quickly the Bragg peaks diminish, they are not visible already at f =8%. At f = 50% the response is featureless and very close to bulk graphite (dashed line in Figure 12a). This is in marked contrast to the a = 20 case, where the Bragg peaks were persistent even at large  $f \approx 50\%$ . In between the two extremes of Figure 12a, however, there is a very interesting region where the scattering just below the diffraction limit (*i.e.*, a little below 2.5 eV) drops to very low values. In particular, for f = 12% there is a photon energy for which total scattering goes to  $\sim$ 7  $\times$  10<sup>-5</sup>, or inversely, the absorption goes above 99.99%. At this point, the incident wave couples strongly, via second order Bragg scattering, to guided resonances within the array. Reflections get minimized, and as the guided resonances are almost parallel to the array surface, they get totally absorbed. Above the  $a = \lambda$  diffraction limit, on the other hand, diffraction in vacuum becomes allowed, that is, a new reflection mode becomes available, and thus the overall reflection sharply increases.

In Figure 12b we plot the absorption of the 10-row, a = 500 nm, f = 12% case, for energies close to the absorption resonance. The background is color-coded for illustration purposes. Most of the green part of the spectrum, between 512-570 nm, has absorption >99%, peaking at >99.99% for  $\lambda = 532$  nm. This makes MWNT arrays an extremely dark material.<sup>31,32,102-107</sup> Such materials have important applications in solar collectors,<sup>108</sup> optical measurement systems,<sup>104</sup> etc. The record holder in terms of darkness is a recently reported low density MWNT system.<sup>31</sup> It is interesting that while both systems, ours and that of ref 31, are nanotube-based, they utilize a different mechanism: here we propose that second order Bragg scattering resonantly redirects the incident light in the film so it gets quickly absorbed, while ref 31 uses a low density MWNT array to index-match the vacuum and thus achieve minimum reflectance. Unavoidable in the latter approach, however, is that total absorption requires thicker films:<sup>32</sup> our estimated absorption

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coefficient is  $\alpha \approx 1 \ \mu m^{-1}$ , about 1 order of magnitude larger than the  $\alpha = 0.12 \ \mu m^{-1}$  of ref 31, making our thickness requirement an order of magnitude smaller. Furthermore, our normal-incidence reflectance is around  $\sim 5 \times 10^{-5}$ , almost an order of magnitude less than ref 31. On the other hand, ref 31 has a wider angular and wavelength acceptance. A complete comparison between the two methods requires additional calculations for p-polarization, variable-angle in-plane and out-of-plane incidence, roughened film surfaces, *etc.*, beyond the scope of this work.

We believe our results to be quite general. Small discrepancies between the actual dielectric function of a given MWNT and the graphite model used here will not alter the main qualitative features we have obtained, that is, the existence of strong Bragg multiple scattering in deep UV frequencies, or strong absorption at optical frequencies. This is also not expected to qualitatively vary under structural changes, that is, other MWNT arrangements such as hexagonal or small structural disorder (in radius, position, and alignment), which may only slightly reduce the size and effectiveness of the band gaps. We can thus use the results and methodologies presented here in conjunction with experiments to (a) assess the nanotube dielectric function from the fine differences in response, and (b) assist in designing new nanotube-based photonic applications such as UV mirrors, waveguides, and lenses, or efficient optical absorbers. On the other hand, we can use our calculations to assess the features of a given MWNT film from its reflectance response, such as MWNT alignment, arrangement, spacing, diameter, disorder, etc.

In conclusion, we calculated the expected response of MWNT arrays under TM illumination, assuming the dielectric response of graphite. We find that strong Bragg scattering with full photonic band gaps can be observed at UV energies. Typical size and structure parameters for this are 20-30 nm nanotube spacing and 12-20 nm diameters. These are within the range of



Figure 12. (a) Normal scattering (reflection + transmission) from 10 rows: a = 500 nm and different *f*. The dashed line marks bulk graphite reflectivity. (b) Normal absorption from 10 rows: a = 500nm; f = 12%. Total absorption (>99.99%) is achieved for  $\lambda = 532$ nm, while the bandwidth of larger than 99% absorption extends through most of the green spectrum.

what can be achieved today, making nanotubes a natural candidate for deep UV photonics. At larger spacing, the Bragg condition red-shifts and gets modulated by the two graphite plasmons resulting into a more complicated response involving band anticrossing and splitting. Finally, at inter-MWNT spacings of the order of 150 nm or more, the Bragg peak shifts to optical frequencies, where the system is more suitable for total absorption. These results will help in designing future nanotube-based Vis-UV photonic applications.

# METHODS

To calculate the photonic response we use the finitedifference time-domain (FDTD) method,<sup>90</sup> in which Maxwell's equations are solved on a computational grid,

$$abla imes \mathbf{E} = -\mu \partial_t \mathbf{H}, \quad \nabla imes \mathbf{H} = \varepsilon_0 \partial_t \mathbf{E} + \partial_t \mathbf{P}_0 + \sum_{m=1}^M \partial_t \mathbf{P}_m$$
 (2)

where material polarization is taken into account through the polarizabilities  $\vec{P}$ 

$$\partial_t^2 \mathbf{P}_0 + \partial_t \mathbf{P}_0 / \tau = \omega_p^2 \varepsilon_0 \mathbf{E}, \quad \partial_t^2 \mathbf{P}_m + \gamma_m \partial_t \mathbf{P}_m + \omega_m^2 \mathbf{P}_m = \sigma_m^2 \varepsilon_0 \mathbf{E}$$
(3)

We calculate (i) reflectivity and effective medium properties and (ii) photonic band structure. For the former we start by solving the transmission problem of a single array row, as shown in Figure 3a (where the CNT is periodically repeated in the vertical direction because of the periodic boundary condition). Two calculations are done, one with and one without tubes, and the complex transmitted and reflected amplitudes t and r (at the dotted lines) are obtained by taking the appropriate ratios. As long as we remain below the diffraction limit, we can use these to calculate the effective medium quantities as follows:<sup>91</sup> from t, r we get the corresponding transfer matrix  $M = [t - r \cdot r/t, r/t; -r/t, 1/t]$ . which we then evaluate in the semi-infinite layer limit  $M^{\infty}$  =  $\lim_{n\to\infty} M^n$  (usually  $n \approx 10^4$ ). This gives us the semi-infinite reflection  $r_{\infty} = -M_{21}^{\infty}/M_{22}^{\infty}$ , which in turn is related through the Fresnel equations to an effective material impedance  $\tilde{z} = (r_{\infty})$  $(r_{\infty} - 1)$ . The trace of the transfer matrix is related to an effective material index  $\tilde{n}$ : Tr{M} = 2 cos( $\tilde{n}\omega a/c$ ), where a is the inter-MWNT spacing. From these we get the effective permittivity and permeability  $\tilde{\epsilon}(\omega) = \tilde{n}(\omega)/\tilde{z}(\omega)$  and  $\tilde{\mu}(\omega) =$  $\tilde{n}(\omega) \cdot \tilde{z}(\omega)$ , which we use to analytically calculate the reflection and transmission from any number of MWNT array rows.

For the photonic band structure, the computational cell consists of a single unit cell, as shown in Figure 3b, where the CNT is periodically repeated in both vertical and horizontal directions because of the periodic boundary conditions (*i.e.*, we simulate a "bulk" 2D array). Bloch periodic boundary conditions are applied corresponding to a particular wavevector **k** along the perimeter of the irreducible Brillouin zone. A point source and monitor are placed in low-symmetry points of the cell. The point source emits a broadband spectrum and we record the time series of the electric field values at the monitor. The peaks of the Fouriertransformed electric field correspond to the eigenfrequencies of the particular **k**. Repeating for all **k**'s we obtain the photonic band structure.<sup>92</sup>

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