# **An investigation of plasma chemistry for dc plasma enhanced chemical vapour deposition of carbon nanotubes and nanofibres**

# **David B Hash**<sup>1</sup>**, Martin S Bell**<sup>2</sup>**, Kenneth B K Teo**<sup>2</sup>**, Brett A Cruden**<sup>1</sup>**, William I Milne**<sup>2</sup> **and M Meyyappan**<sup>1</sup>*,*<sup>3</sup>

<sup>1</sup> Center for Nanotechnology, NASA Ames Research Center, Moffett Field, CA 94035, USA <sup>2</sup> Department of Engineering, University of Cambridge, Cambridge CB2 1PZ, UK

E-mail: M.Meyyappan@nasa.gov

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#### **Abstract**

The role of plasma in plasma enhanced chemical vapour deposition of carbon nanotubes and nanofibres is investigated with both experimental and computational diagnostic techniques. A residual gas analysis (RGA) of a 12 mbar dc discharge with a  $C_2H_2/NH_3$  gas mixture is conducted near the Ni catalyst surface employed for carbon nanofibre growth. The results are corroborated with a 1D dc discharge model that solves for species densities, ion momentum, and ion, electron and neutral gas thermal energies. The effect of varying the plasma power from 0 to 200 W on the gas composition is studied. The dissociation efficiency of the plasma is demonstrated where over 50% of the feedstock is converted to a mixture of hydrogen, nitrogen and hydrogen cyanide at 200 W. Finally, the important role that endothermic ion–molecule reactions play in this conversion is, for the first time, established. Of these reactions, dissociative proton abstraction and collision-induced dissociation are of the greatest significance.

(Some figures in this article are in colour only in the electronic version)

# **1. Introduction**

Catalytic decomposition of carbon-bearing gases on metallic nanocluster catalysts in chemical vapour deposition (CVD) systems for growth of carbon nanotubes and nanofibres combines the advantages of low temperature (relative to arc discharge and laser ablation), low cost, commercial scale production with the ability of pattern growth through lithographic positioning of transition metal catalysts on substrates. Many potential applications of nanotubes such as atomic force microscope tips [1], superhydrophobic surfaces [2], field emission cathodes [3, 4], vertical interconnects [5, 6], electron beam lithography [7–10], synthetic membranes [11, 12], intracellular gene delivery devices [13, 14], and nanoelectrode electrochemical probes

alignment where carbon nanotubes are grown closely together like towers or through template-assisted growth. However, the application of large electric fields has been shown to provide superior alignment [20–24] by inducing dipole moments preferentially along the axes of carbon nanotubes that act to align the tube in the direction of the field and combat any randomizing effect of thermal vibrations. Electric field enabled alignment is exploited in plasma enhanced CVD with various configurations including dc [25–31], rf [32–35], microwave [36–38], inductive [39–44] and electron cyclotron resonance [45–47]. Lin *et al* [48] demonstrated that the alignment mechanism is the same for all systems. Regardless of whether a system employs a microwave plasma with a selfbias of 10 V and sheath width of the order of 100 *µ*m or

and biosensors [15–19] require not only patterned growth but also vertical alignment. Alignment in thermal CVD processes can be obtained through van der Waals interaction sponsored a dc discharge with 600 V applied bias and 2 mm sheath width, the sheath electric fields are of the same order and are requisite for aligned growth. The plasma itself, in fact, is not required for aligned growth as [21] and [22] employed fields through small applied voltages (3–20 V) across very small electrode spacings  $(10-100 \mu m)$  avoiding striking a discharge. This demonstrates that the only requirement for aligned growth is an electric field and that it is the sheath electric fields of the plasma and not the plasma itself that affects alignment. However, large-scale electric field aligned nanotube production has yet to be accomplished without a plasma; and thus, as it is at present unavoidable for largescale superiorly aligned growth, it is important to understand the comprehensive role of the plasma. Specifically, the present work examines the effect of the plasma chemistry on gas composition in dc PECVD and thus the subsequent precursors to nanotube growth. The investigation relies on both experimental and computational methods to perform plasma diagnostics. Within the computational approach, the importance of modelling of sheath endothermic ion–molecule reactions is examined. The experimental approach involves a residual gas analysis (RGA) study of the plasma and adds to the growing but still limited work in this area [49–53].

## **2. CNT Growth**

The reactor employed for this study is a simple dc configuration of equal area  $(10 \text{ cm}^2)$  cathode and anode with a 5 cm separation. The feedstock of 54:200 sccm of  $C_2H_2/NH_3$ is injected through a shower head, which also acts as the anode. The graphite cathode has an embedded rigid tungsten wire heater coupled with an electrically isolated thermocouple to allow independent temperature control of the substrate if necessary.

 $Si(100)$  substrates were coated with conductive indium tin oxide (15 nm thick) and Ni (7 nm) thin films by magnetron sputtering. The substrate was placed on the cathode and a dc glow discharge was initiated at low power and pressure  $(20 \text{ W}, 2.5 \text{ mbar})$  in pure NH<sub>3</sub>. The power and pressure were then simultaneously increased to 120 W and 12 mbar, respectively, and a cathode temperature of 550 ◦C was typically obtained after just 1 min. This catalyst pretreatment procedure transformed the Ni thin film into nanoclusters of the range between 50 and 100 nm. After the 1 min NH<sub>3</sub> plasmaannealing step,  $C_2H_2$  was introduced into the gas mixture and growth was performed at the desired plasma power for 15 min. All cases reported here were performed at a pressure of 12 mbar, and the desired plasma power in the range 20– 200 W was achieved by varying the applied dc bias from 470 to 650 V. The substrate was heated solely by the plasma, and its temperature varied between 350 and 715 ◦C over the power range investigated though the low range was not suitable for growth.

#### **3. Computational model**

The role of the plasma is investigated with a 1D radially averaged computational model [54]. Equations for the conservation of species mass, ion momentum, and ion, electron and neutral gas thermal energy are solved axially between the



two electrodes. The model includes 21 neutral species  $(H_2,$ H, CH4, CH3, CH2, CH, C2H4, C2H3, C2H2, C2H, N2, N,  $NH_3$ , NH<sub>2</sub>, NH, NNH, HCN, CN, HC<sub>3</sub>N, H<sub>2</sub>CN, H<sub>2</sub>CNH), 7 charged species (NH<sub>3</sub>, NH<sub>2</sub>, C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H<sup>+</sup>, H<sub>2</sub><sup>+</sup>, H<sup>+</sup>, e), and 200 reactions. As a boundary condition for the gas energy equation, a cathode energy balance is incorporated to model ion bombardment, thermal radiation and solid and gas conduction to predict the cathode temperature.

The computational model includes endothermic ion– molecule reactions that have not previously been investigated in simulations. Inclusion of these reactions requires the addition of the ion momentum and energy equations to compute both the directed and thermal energies of the ions. Endothermic ion–molecule reactions only occur in the sheath where ions have energies larger than the reaction barrier. Studies of low-pressure discharges for semiconductor applications, in which sheaths are collisionless, have often neglected endothermic ion–molecule reactions while including exothermic ones. Peko *et al* [55] has advocated for the inclusion of ion–molecule reactions in simulations, as they are 'essential because of the important role that secondary products from ion–molecule reactions play in the etching and deposition processes'. This is especially true in nanotube processing where sheaths are collisional because of higher operating pressures (∼10 mbar) and ion energies are significant because of the large applied biases (∼500 V) required for alignment such that both exothermic and endothermic ion–molecule reactions are important. Table 1 displays the rates employed for exothermic ion–molecule reactions, and the endothermic rates for charge transfer, dissociative charge transfer, dissociative proton abstraction and collision-induced dissociation are given in table 2. Rates for the exothermic reactions are easily found in the modelling literature of planetary atmospheres; however, rates of endothermic ion–molecule reactions for most nanotube processing gases are not available. As a result, the rates of

<b>Table 2.</b> Endothermic ion–molecule reactions.		
Reaction	Rate $\rm (cm^3~s^{-1})$	Reference
Charge transfer		
$H^+ + H_2 \rightarrow H_2^+ + H$ $C_2H_2^* + H_2 \rightarrow H_2^* + C_2H_2$ $NH3+ + H2 \rightarrow H3+ + NH3$	$1.57 \times 10^{-9} T^{-0.13} \exp(-2.9/T)$ $4.0 \times 10^{-9} T^{-0.25}$ exp(-14.6/T) $4.0 \times 10^{-9} T^{-0.25}$ exp(-14.6/T)	[67] $[56]$ <sup>a</sup> $[56]$ <sup>a</sup>
Collision-induced dissociation		
$C_2H_2^+ + C_2H_2 \rightarrow H^+ + C_2H + C_2H_2$ $NH_3^+ + NH_3 \rightarrow H^+ + NH_2 + NH_3$ $C_2H_2^* + H_2 \rightarrow H^* + C_2H + H_2$ $NH_3^+ + H_2 \rightarrow H^+ + NH_2 + H_2$	$1.81 \times 10^{-9} T^{-0.17} \exp(-4.6/T)$ $1.81 \times 10^{-9} T^{-0.17} \exp(-4.6/T)$ $6.8 \times 10^{-9} T^{-0.2} \exp(-9.9/T)$ $6.8 \times 10^{-9} T^{-0.2} \exp(-9.9/T)$	$[56]$ <sup>a</sup> $[56]$ <sup>a</sup> $[56]$ <sup>a</sup> $[56]$ <sup>a</sup>
Dissociative proton abstraction		
$C_2H_2^+ + C_2H_2 \rightarrow C_2H^+ + H_2 + C_2H$ $NH_3^+ + NH_3 \rightarrow NH_2^+ + H_2 + NH_2$	$7.5 \times 10^{-8} T^{-0.44}$ exp(-3.8/T) $7.5 \times 10^{-8} T^{-0.44} \exp(-3.8/T)$	$[56]$ <sup>a</sup> $[56]$ <sup>a</sup>
Dissociative charge transfer		
$H_2^+ + H_2 \rightarrow H^+ + H + H_2$ $C_2H_2^+ + C_2H_2 \rightarrow C_2H^+ + H + C_2H_2$ $NH_3^+ + NH_3 \rightarrow NH_2^+ + H + NH_3$ $C_2H_2^* + H_2 \rightarrow H^* + H + C_2H_2$ $NH_3^+ + H_2 \rightarrow H^+ + H + NH_3$	$1.97 \times 10^{-9} T^{-0.19} \exp(-5.3/T)$ $3.7 \times 10^{-9} T^{-0.3} \exp(-21.1/T)$ $3.7 \times 10^{-9} T^{-0.3} \exp(-21.1/T)$ $7.8 \times 10^{-9} T^{-0.39}$ exp(-14.8/T) $7.8 \times 10^{-9} T^{-0.39}$ exp(-14.8/T)	[67] $[56]$ <sup>a</sup> $[56]$ <sup>a</sup> $[56]$ <sup>a</sup> $[56]$ <sup>a</sup>

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 $a$  Estimate from CH<sub>4</sub>.



**Figure 1.** Per cent ammonia decomposition 1 cm from the cathode. The circles are experimental measurements, the squares are the full simulation results including all ion–molecule reactions, and the triangles are the simulation results excluding endothermic ion–molecule reactions.

these reactions for methane from the work of Peko *et al* [56] were used as estimates for acetylene and ammonia, where the experimentally measured cross-sections reported for these reactions were integrated over a Maxwellian distribution and then fitted to the Arrhenius form.

#### **4. Experimental diagnostics**

Mass spectrometry was performed 1 cm from the cathode using a Hiden EQP High Energy plasma analyser differentially pumped to  $1 \times 10^{-6}$  mbar. For neutral species measurements, the acquired mass spectra must be deconvolved as the resulting intensities, *i*, from the RGA are products of the original species cracking patterns. For a system of *n* species and *m* spectra, the following matrix [57] must be solved for [*I*]



**Figure 2.** Per cent acetylene decomposition at 1 cm from the cathode. The circles are experimental measurements, the squares are the full simulation results including all ion–molecule reactions, and the triangles are the simulation results excluding endothermic ion–molecule reactions.

$$
\begin{bmatrix} i_1 \\ i_2 \\ \bullet \\ i_m \end{bmatrix} = \begin{bmatrix} a_{11} & \bullet & \bullet & a_{1n} \\ a_{21} & \bullet & \bullet & a_{2n} \\ \bullet & \bullet & \bullet & \bullet \\ a_{m1} & \bullet & \bullet & a_{mn} \end{bmatrix} \begin{bmatrix} I_1 \\ I_2 \\ \bullet \\ I_n \end{bmatrix} \tag{1}
$$

where *a* represents the cracking patterns for each of the *n* species. The 70 eV cracking patterns are taken from the NIST Chemistry WebBook [58]. To avoid overfitting the data, *n* was limited to 17 species, which best fit the data and were consistent with the modelling results.

## **5. Results and discussion**

Figures 1 and 2 display the per cent decomposition of the feedstock gases, ammonia, and acetylene, as a function of



**Figure 3.** Ammonia ion loss rate from endothermic ion–molecule reactions in the cathode sheath. Dissociative proton abstraction and collision-induced dissociation are the dominant reactions.



**Figure 4.** Acetylene ion loss rate from endothermic ion–molecule reactions in the cathode sheath. Dissociative proton abstraction and collision-induced dissociation are the dominant reactions.

plasma power. It is defined as the per cent change in mole fraction relative to the plasma off condition

$$
\frac{X_s^* - X_s}{X_s^*} = \frac{P_s^* / P^* - P_s / P}{P_s^* / P^*}
$$
 (2)

where  $X_s$  is the mole fraction of species  $s$ ,  $*$  denotes the plasma off condition, and  $P_s$  is the partial pressure of species *s*. Given that the total pressure is maintained constant for all the cases considered and the intensities from the RGA are proportional to the partial pressures, equation (2) can be rewritten as

$$
\frac{X_s^* - X_s}{X_s^*} = \frac{I_s^* - I_s}{I_s^*}.
$$
 (3)

In the figures, three separate sets of data are displayed: the RGA measurements, the simulation with the full reaction set, and the simulation with a reduced reaction set excluding the endothermic ion–molecule reactions. For ammonia, the comparison between the experiment and the full simulation is good between 0 and 100 W but suffers at high powers. For acetylene, even the full simulation is not able to reproduce the experimental result where over 80% of the feedstock acetylene is decomposed at the highest plasma power. Possible reasons for the discrepancy may include approximations in the RGA



**Figure 5.** Mole fraction trends of main plasma products from simulation and experiment at 1 cm from the cathode.

analysis, rate estimates used for ion–molecule reactions, or loss of carbon to surface reactions. Of these, the rate estimates used are the most probable source of error. Because cross section measurements for endothermic ion–molecule reactions in acetylene and ammonia cannot be found in the literature, the exact same rates for these reactions were employed for both gases. Thus, it is likely that the methane rates employed more accurately represent ammonia ion–molecule reaction rates than that of acetylene, explaining the more accurate comparison to experiment for ammonia than for acetylene. Despite these possible shortcomings, these plots provide an important insight into some of the relevant plasma chemistry, namely the importance of incorporating endothermic ion– molecule reactions into the model. The per cent difference between the simulations with and without these reactions is quite significant, ranging from 50 to 200% with the full simulations providing a better comparison to the experiment for both ammonia and acetylene. Figures 3 and 4 display the ion loss rate for each of the endothermic ion–molecule reactions employed for ammonia and acetylene across the electrode spacing. The rate axis is plotted logarithmically, dramatically demonstrating that these reactions occur solely in the 1.5 mm sheath adjacent to the cathode. For both ammonia and acetylene, the dominant endothermic ion– molecule reaction is dissociative proton abstraction followed by collision-induced dissociation. At least these two reaction mechanisms should be included in simulations of dc discharges for nanotube processing to obtain reliable results.

Figure 5 shows the mole fraction trends with plasma power for the three main plasma products:  $H_2$ ,  $N_2$  and HCN. To compare the experiment with the simulations, the experimental data are calibrated to the simulation at the 100 W midpoint. The simulation results displayed here are from the full reaction set employing all ion–molecule reactions. The trends are reproduced well where hydrogen and hydrogen cyanide increase with increasing power and nitrogen remains relatively flat. The exception is HCN at lower powers where the simulation shows a much larger increase in HCN than the experiment. HCN is mainly formed from the reaction  $CH + HCN \Leftrightarrow N + C_2H_2$  whose equilibrium constant derived backward reaction rate in the simulation was calculated from a published [59] estimated forward reaction rate of 1*.*66 ×  $10^{-10}$  cm<sup>3</sup> s<sup>-1</sup>. The region of discrepancy between 0 and



**Figure 6.** Simulated neutral species in the gas phase at 200 W plasma power plotted versus the distance between the electrodes. The seven most dominant stable species with the three most significant radical species are shown here.

50 W has a significant gas temperature rise, and it may be that the reaction in actuality has a temperature dependence that would mitigate the differences between the experiment and simulation.

Figure 6 illustrates the significant role plasma plays in nanotube processing. The simulated mole fractions of the seven most dominant stable species are shown with the three most significant radicals over the electrode spacing at the highest plasma power investigated (200 W). The ammonia and acetylene feedstock gases dissociate near the cathode in the high-energy sheath region resulting in the highest concentration of radicals. At this power, the plasma can convert enough of the acetylene into hydrogen cyanide to make it the most dominant carbon-bearing species at the cathode. Figure 7 is a similar plot in which the species number densities at the cathode where the substrate is positioned are plotted versus plasma power. Here, it is clear that the growth sample will see more HCN than  $C_2H_2$ . The dramatic impact this has on growth has been previously demonstrated [60], whereby, at the same substrate temperature, higher power conditions result in much shorter nanofibres relative to lower power growth conditions. This was attributed to acetylene decomposing more readily on the Ni catalyst than HCN given that the C–N triple bond strength [61] is 748 kJ mol<sup>-1</sup> and the H–CCH bond strength is only 556 kJ mol<sup>-1</sup>.

#### **6. Conclusions**

The role of plasma in dc plasma enhanced chemical vapour deposition of carbon nanofibres has been investigated both experimentally and computationally. Both demonstrate the significant role plasma plays in determining the gas-phase species impinging on the catalysts. The diagnostics show that a 12 mbar 200 W dc discharge with a 5 cm electrode gap and  $10 \text{ cm}^2$  area efficiently dissociates the ammonia and acetylene feedstock gases by as much as 50–60% and 60– 80%, respectively. The resulting new carbon-bearing species (hydrogen cyanide, methanimine, and methane) will catalyse at different rates than the original feedstock, thus impacting



**Figure 7.** Simulation of the effect of plasma power on gas phase number densities at the substrate where carbon nanotube growth occurs.

final growth rates. Depending on the feedstock gas mixture and the catalyst employed, this feedstock conversion may or may not be advantageous. In our case where Ni catalyst was used, the generation of hydrogen cyanide at the expense of acetylene at higher powers actually impeded the nanotube/fibre growth rate, in contrast to conventional thinking which assumes that higher plasma powers would lead to more efficient/higher growth rates. The result is that the term plasma enhanced CVD becomes a misnomer where the plasma chemistry actually results in decreased growth rates. In order to produce optimal growth conditions with the  $C_2H_2/NH_3$  gas mixture, the plasma power is maintained low enough to reduce the degree of gas conversion while still high enough to guarantee superior alignment from the plasma sheath fields.

For the first time, the important role of endothermic ion– molecule reactions has been demonstrated in plasma-assisted nanotube processing. In many applications, these reactions can be, and have been, ignored as they occur only in the sheaths and require significant pressures and ion energies. The high pressures (∼10 mbar) and high applied dc biases (∼500 V) in nanotube processing, however, present the perfect condition for these reactions, and it is demonstrated here that neglecting them may lead to errors in the prediction of gas phase composition of the order of a factor of two.

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#### **References**

[1] Ye Q, Cassell A M, Liu H, Chao K-J, Han J and Meyyappan M 2004 *Nano Lett.* **4** 1301

- [2] Lau K K S, Bico J, Teo K B K, Chhowalla M, Amaratunga G A J, Milne W I, McKinley G H and Gleason K K 2003 *Nano Lett.* **3** 1701
- [3] Pirio G, Legagneux P, Pribat D, Teo K B K, Chhowalla M, Amaratunga G A J and Milne W I 2002 *Nanotechnology* **13** 1
- [4] Gangloff L *et al* 2004 *Nano Lett.* **4** 1575
- [5] Li J, Ye Q, Cassell A, Ng H T, Stevens R, Han J and Meyyappan M 2003 *Appl. Phys. Lett.* **82** 2491
- [6] Ngo Q, Petranovic D, Krishnan S, Cassell A M, Ye Q, Li J, Meyyappan M and Yang C Y 2004 *IEEE Trans. Nanotechnol.* **3** 311
- [7] Guillorn M A, Melechko A V, Merkulov V I, Ellis E D, Simpson M L, Lowndes D H, Baylor L R and Bordonaro G J 2001 *J. Vac. Sci. Technol.* B **19** 2598
- [8] Baylor L R, Lowndes D H, Simpson M L, Thomas C E, Guillorn M A, Merkulov V I, Wheaton J H, Ellis E D, Hensley D K and Melechko A V 2002 *J. Vac. Sci. Technol.* B **20** 2646
- [9] Teo K B K *et al* 2003 *J. Vac. Sci. Technol.* B **21** 693
- [10] Baylor L R *et al* 2004 *J. Vac. Sci. Technol.* B **22** 3021
- [11] Zhang L, Melechko A V, Merkulov V I, Guillorn M A, Simpson M L, Lowndes D H and Doktycz M J 2002 *Appl. Phys. Lett.* **81** 135
- [12] Fletcher B L, Hullander E D, Melechko A V, McKnight T E, Klein K L, Hensley D K, Morrell J L, Simpson M L and Doktycz M J 2004 *Nano Lett.* **4** 1809
- [13] McKnight T E, Melechko A V, Griffin G D, Guillorn M A, Merkulov V I, Serna F, Hensley D K, Doktycz M J, Lowndes D H and Simpson M L 2003 *Nanotechnol.* **14** 551
- [14] McKnight T E, Melechko A V, Hensley D K, Mann D G J, Griffin G D and Simpson M L 2004 *Nano Lett.* **4** 1213
- [15] Guillorn M A, McKnight T E, Melechko A, Merkulov V I, Britt P F, Austin D W, Lowndes D H and Simpson M L 2002 *J. Appl. Phys.* **91** 3824
- [16] Li J, Ng H T, Cassell A, Fan W, Chen H, Ye Q, Koehne J, Han J and Meyyappan M 2003 *Nano Lett.* **3** 597
- [17] Koehne J, Chen H, Li J, Cassell A M, Ye Q, Ng H T,
- Han J and Meyyappan M 2003 *Nanotechnology* **14** 1239 [18] Lin Y, Lu F, Tu Y and Ren Z 2004 *Nano Lett.* **4** 191
- [19] McKnight T E, Melechko A V, Austin D W, Sims T,
- Guillorn M A and Simpson M L 2004 *J. Phys. Chem.* B **108** 7115
- [20] Avigal Y and Kalish R 2001 *Appl. Phys. Lett.* **78** 2291
- [21] Zhang Y, Chang A, Cao J, Wang Q, Kim W, Li Y, Morris N, Yenilmez E, Kong J and Dai H 2001 *Appl. Phys. Lett.* **79** 3155
- [22] Ural A, Li Y and Dai H 2002 *Appl. Phys. Lett.* **81** 3464
- [23] Delzeit L, Stevens R, Nguyen C and Meyyappan M 2002 *Int. J. Nanosci.* **1** 197
- [24] Joselevich E and Lieber C M 2002 *Nano Lett.* **2** 1137
- [25] Ren Z F, Huang Z P, Xu J W, Wang J H, Bush P,
- Siegal M P and Provencio P N 1998 *Science* **282** 1105 [26] Teo K B K, Chhowalla M, Amaratunga G A J, Milne W I,
- Hasko D G, Pirio G, Legagneux P, Wyczisk F and Pribat D 2001 *Appl. Phys. Lett.* **79** 1534
- [27] Chhowalla M, Teo K B K, Ducati C, Rupesinghe N L, Amaratunga G A J, Ferrari A C, Roy D, Robertson J and Milne W I 2001 *J. Appl. Phys.* **90** 5308
- [28] Merkulov V I, Melechko A V, Guillorn M A, Simpson M L, Lowndes D H, Whealton J H and Raridon R J 2002 *Appl. Phys. Lett.* **80** 4816
- [29] Yang Q, Xiao C, Chen W, Singh A K, Asai T and Hirose A 2003 *Diamond Relat. Mater.* **12** 1482
- [30] Täschner Ch, Pácal F, Leonhardt A, Spatenka P, Bartsch K, Graff A and Kaltofen R 2003 *Surf. Coat. Technol.* **174/175** 81
- [31] AuBuchon J F, Chen L-H, Gapin A I, Kim D-W, Daraio C and Jin S 2004 *Nano Lett.* **4** 1781
- [32] Ikuno T, Furuta H, Yamamoto T, Takahashi S, Kamizono M, Honda S, Katayama M, Hirao T and Oura K 2003 *Surf. Interface Anal.* **35** 15
- [33] Kato T, Jeong G-H, Hirata T, Hatakeyama R, Tohji K and Motomiya K 2003 *Chem. Phys. Lett.* **381** 422
- [34] Kato T, Jeong G-H, Hirata T and Hatakeyama R 2004 *Thin Solid Films* **457** 2
- [35] Li Y *et al* 2004 *Nano Lett.* **4** 317
- [36] Tsai S H, Chao C W, Lee C L and Shih H C 1999 *Appl. Phys. Lett.* **74** 3462
- [37] Bower C, Zhu W, Jin S and Zhou O 2000 *Appl. Phys. Lett.* **77** 830
- [38] Chen M, Chen C-M and Chen C-F 2002 *Thin Solid Films* **420/421** 230
- [39] Delzeit L, McAninch I, Cruden B A, Hash D, Chen B, Han J and Meyyappan M 2002 *J. Appl. Phys.* **91** 6027
- [40] Delzeit L, Nguyen C V, Stevens R M, Han J and Meyyappan M 2002 *Nanotechnology* **13** 280
- [41] Matthews K, Cruden B A, Chen B, Meyyappan M and Delzeit L 2002 *J. Nanosci. Nanotechnol.* **2** 475
- [42] Caughman J B O, Baylor L R, Guillorn M A, Merkulov V I, Lowndes D H and Allard L F 2003 *Appl. Phys. Lett.* **83** 1207
- [43] Honda S, Katayama M, Lee K, Ikuno T, Ohkura S, Oura K, Furuta H and Hirao T 2003 *Japan. J. Appl. Phys.* **42** L441
- [44] Lee K, Katayama M, Honda S, Kuzuoka T, Miyake T, Terao Y, Lee J, Mori H, Hirao T and Oura K 2003 *Japan. J. Appl. Phys.* **42** L804
- [45] Hsu C M, Lin C H, Chang H L and Kuo C T 2002 *Thin Solid Films* **420/421** 225
- [46] Woo Y S, Jeon D Y, Han I T, Park Y J, Kim H J, Jung J E, Kim J M and Lee N S 2003 *J. Vac. Sci. Technol.* B **21** 1660
- [47] Lin C H, Lee S H, Hsu C M and Kuo C T 2004 *Diamond Relat. Mater.* **13** 2147
- [48] Lin C C, Leu I C, Yen J H and Hon M H 2004 *Nanotechnology* **15** 176
- [49] Woo Y S, Jeon D S, Han I T, Lee N S, Jung J E and Kim J M 2002 *Diamond Relat. Mater.* **11** 59
- [50] Cruden B A, Cassell A M, Ye Q and Meyyappan M 2003 *J. Appl. Phys.* **94** 4070
- [51] Denysenko I B, Xu S, Long J D, Rutkevych P P, Azarenkov N A and Ostrikov K 2004 *J. Appl. Phys.* **95** 2713
- [52] Bell M S, Lacerda R G, Teo K B K, Rupesinghe N L, Amaratunga G A J, Milne W I and Chhowalla M 2004 *Appl. Phys. Lett.* **85** 1137
- [53] Cruden B A, Cassell A M, Hash D B and Meyyappan M 2004 *J. Appl. Phys.* **96** 5284
- [54] Hash D, Bose D, Govindan T R and Meyyappan M 2003 *J. Appl. Phys.* **93** 6284
- [55] Peko B L, Champion R L, Rao M V V S and Olthoff J K 2002 *J. Appl. Phys.* **92** 1657
- [56] Peko B L, Dyakov I V and Champion R L 1998 *J. Chem. Phys.* **109** 5269
- [57] O'Hanlon J F 1980 *A User's Guide to Vacuum Technology* (New York: Wiley) pp 105–19
- [58] *NIST Chemistry WebBook* <http://webbook.nist.gov>
- [59] Dean A J, Hanson R K and Bowman C T 1991 *J. Phys. Chem.* **95** 3180
- [60] Teo K B K *et al* 2004 *Nano Lett.* **4** 921
- [61] Lide D R (ed) 2000 *CRC Handbook of Chemistry and Physics* 81st edn (Boca Raton, FL: CRC Press)
- [62] Anicich V G 1993 *J. Phys. Chem. Ref. Data* **22** 1469
- [63] Operti L, Rabezzana R and Vaglio G A 2003 *Int. J. Mass Spectrom.* **228** 403
- [64] Operti L, Rabezzana R, Turco F and Vaglio G A 2004 *Int. J. Mass Spectrom.* **232** 139
- [65] Anicich V G and McEwan M J 1997 *Planet. Space Sci.* **45** 897
- [66] Keller C N, Cravens T E and Gan L 1992 *J. Geophys. Res.* **97** 12117
- [67] Phelps A V 1990 *J. Phys. Chem. Ref. Data* **19** 653